

An Introduction to Multi-Anvil Techniques

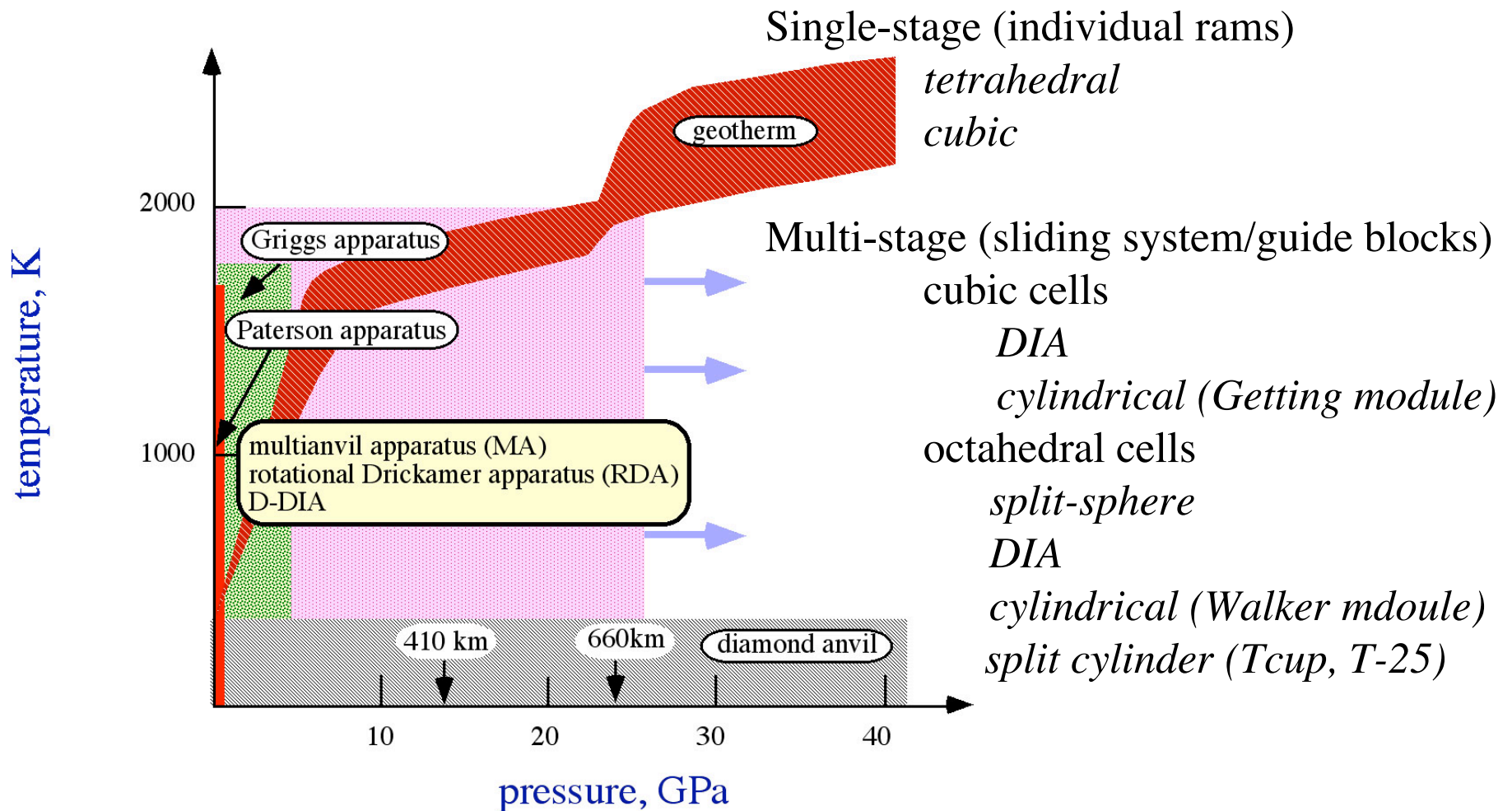
Charles E Lesher
UC Davis

*2005 COMPRES Multi-Anvil Workshop
March 1-3, GSECARS - APS*

OUTLINE

- Types of multianvil systems
- Design considerations
- Importance of gasketing
- Cell assemblies
- Pressure generation
- Heat generation
- Calibration

Multianvil Systems



Single-stage multianvil systems

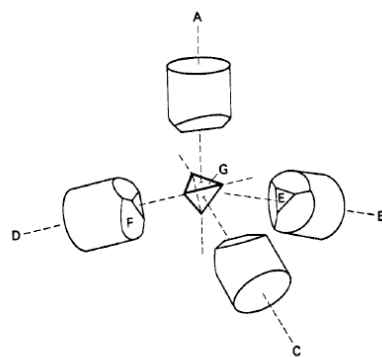


Fig. 2.26. Cluster of tetrahedral anvils.

Single-stage multianvil systems

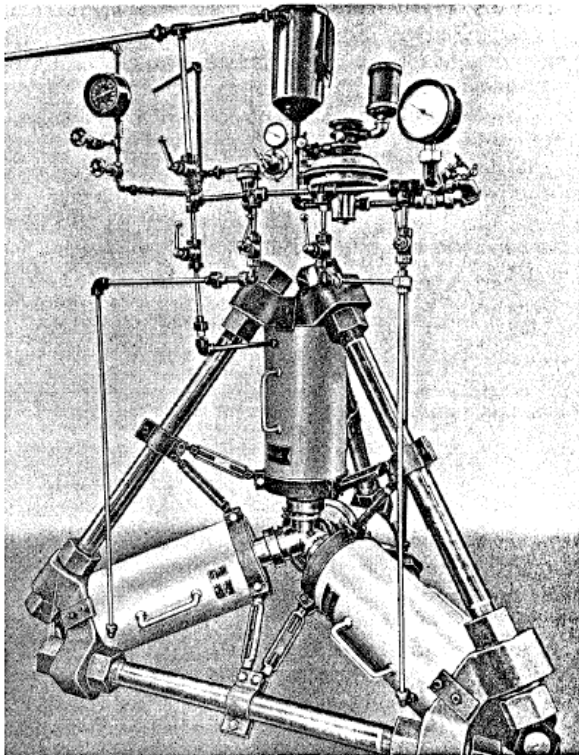


Fig. 2.25. The original tetrahedral press.

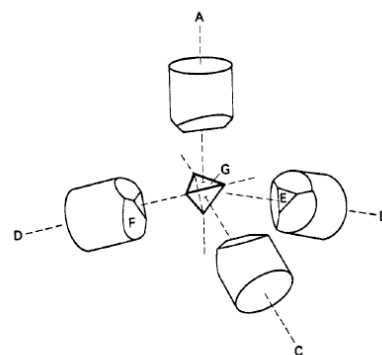


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Single-stage multianvil systems

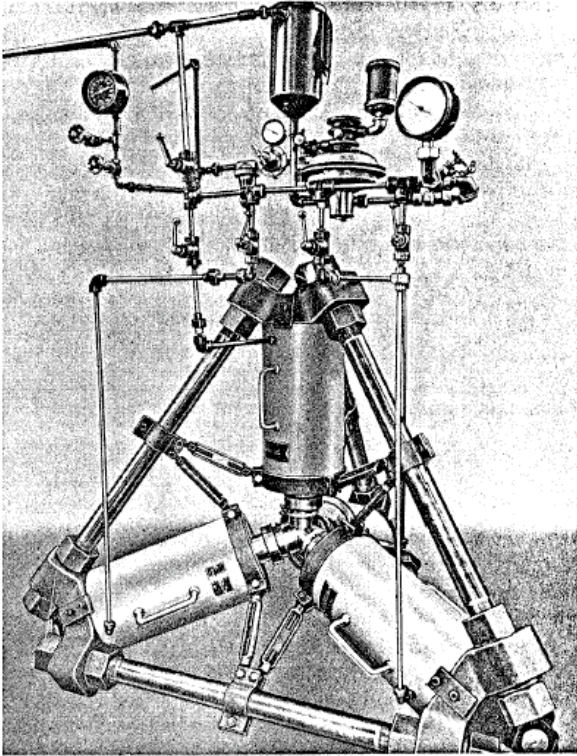


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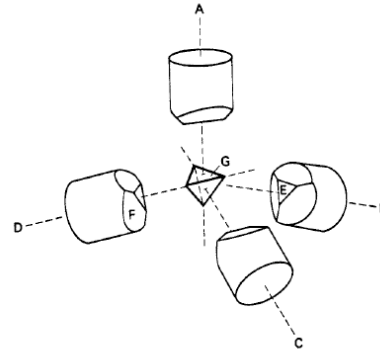
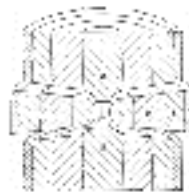


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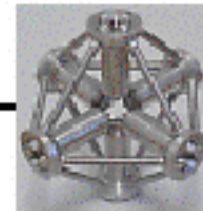
Press Evolution:



Belt Press



Tetrahedral Press



Cubic Press

H. Tracy Hall Foundation

Single-stage multianvil systems

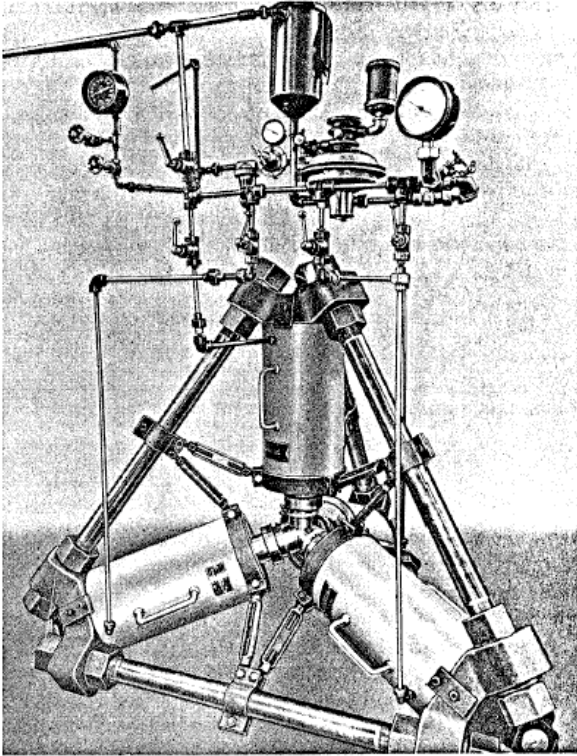


Fig. 2.25. The original tetrahedral press.



NovaDiamonds

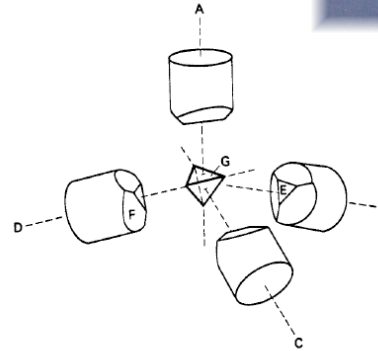
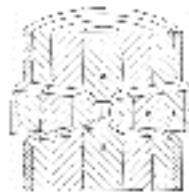


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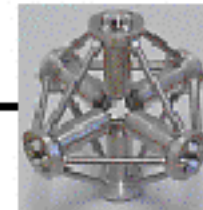
Press Evolution:



Belt Press



Tetrahedral Press



Cubic Press

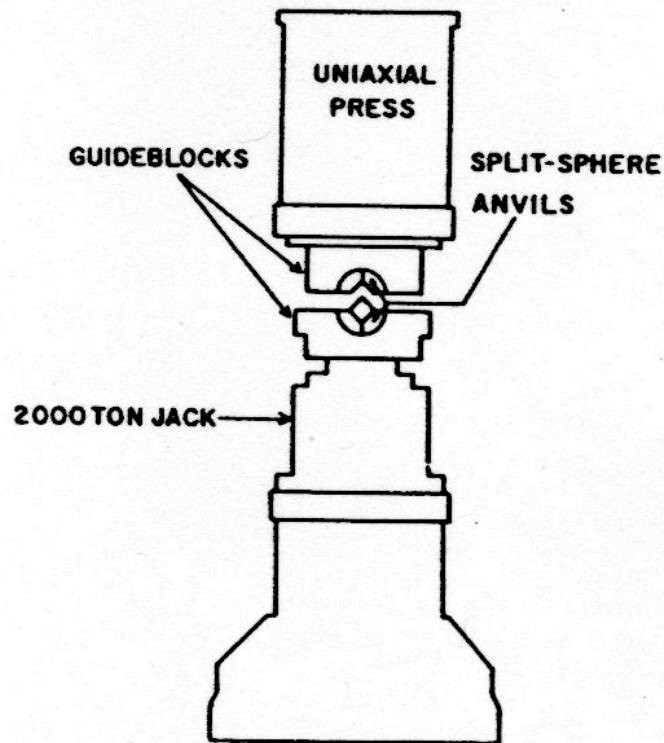


H. Tracy Hall Foundation

Multi-stage multianvil systems

Split-sphere apparatus

UNIAXIAL SPLIT SPHERE APPARATUS
USSA - 2000



Sumitomo 2000-Ton Press (SUNY-Stony Brook)

Kawai- type (6-8)

*First stage:
6 spherical anvils*



*Second stage:
8 truncated cubic anvils*

USSA - 2000

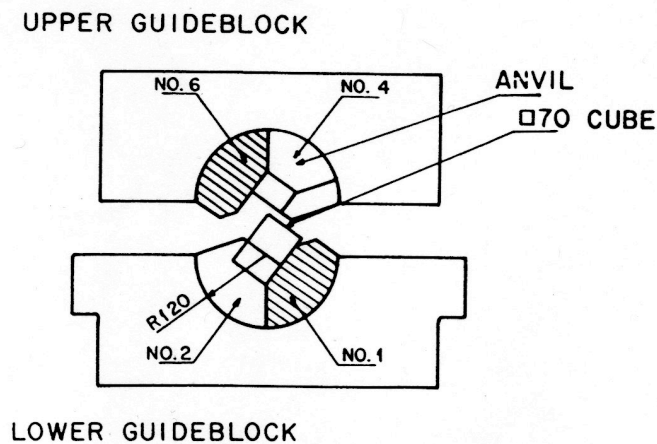


Figure 1

Schematic diagrams for 2000-ton uniaxial split-sphere apparatus (USSA-2000). (a) Hydraulic press with guideblocks and split-sphere anvils in place. (b) Details of the first-stage spherical steel anvils in the guideblocks. (c) Second-stage with the assembly of eight tungsten carbide cubic anvils of truncation edge length (TEL = a) which compress an octahedral pressure medium.

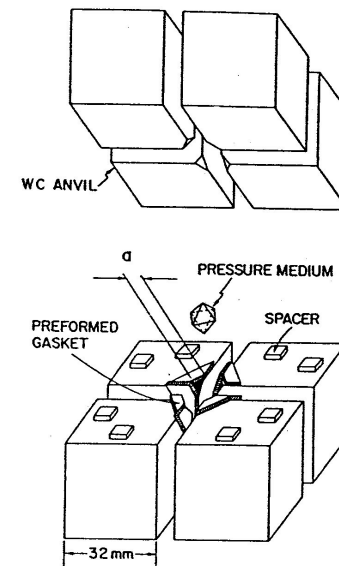
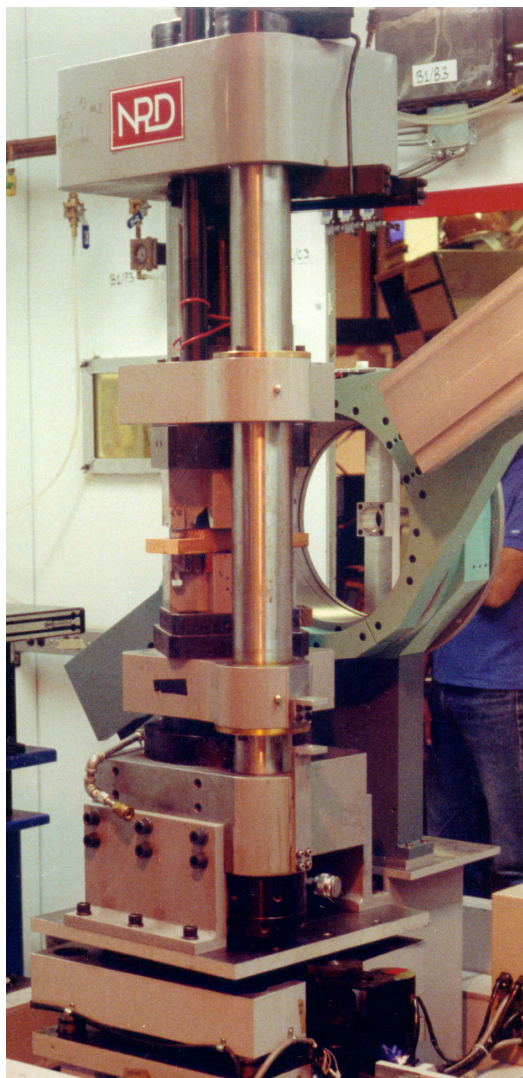


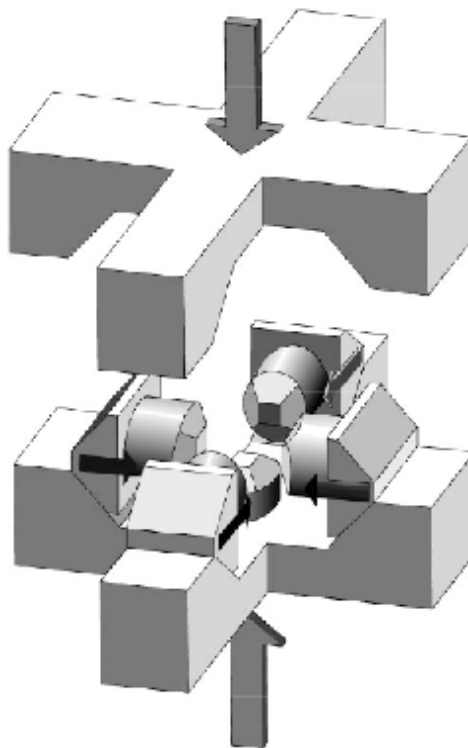
Figure 4. Second stage of the USSA-2000 with the assembly of eight tungsten carbide cubic anvils of truncation edge length (a) which compress an octahedral pressure medium (modified after Sawamoto, 1986).

DIA apparatus

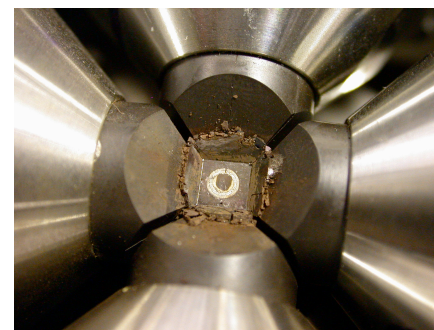


**SAM85
at X17B2 (NSLS)**

(a)

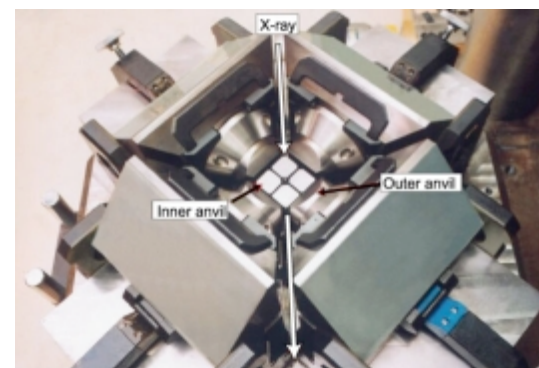


8-6



Cubic cell

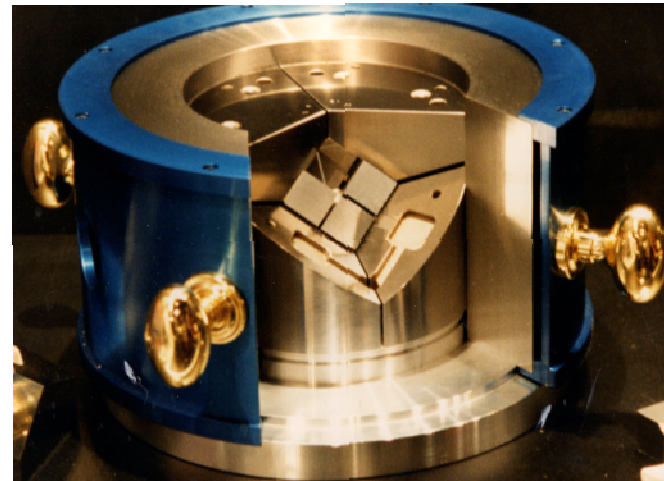
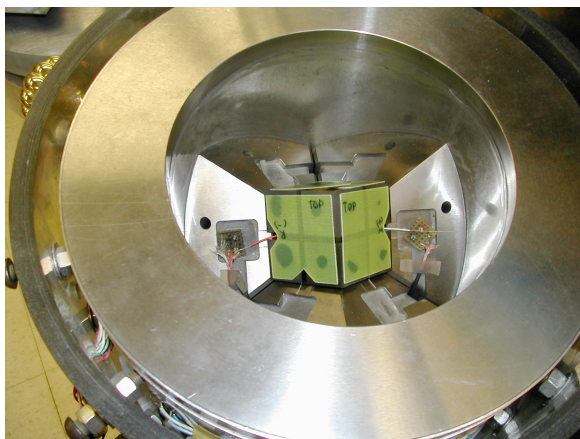
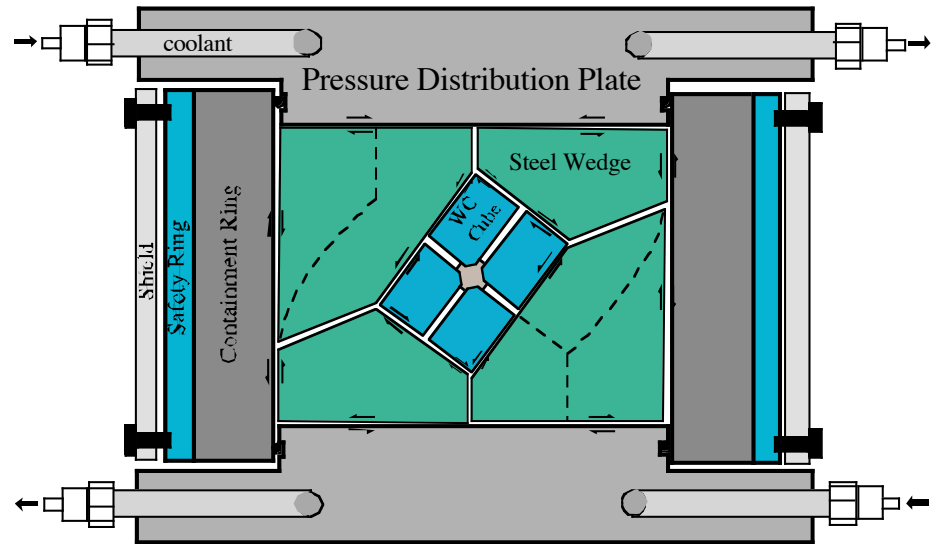
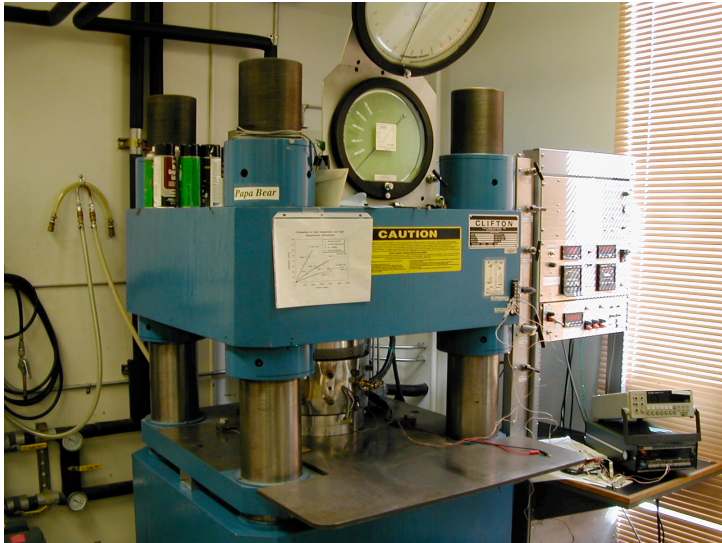
8-6-8



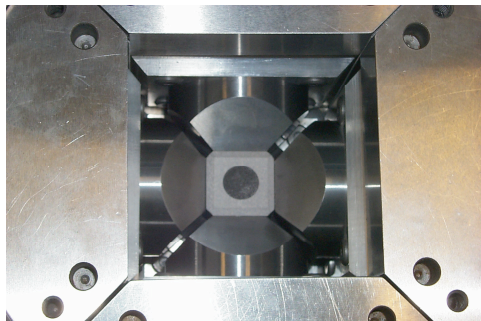
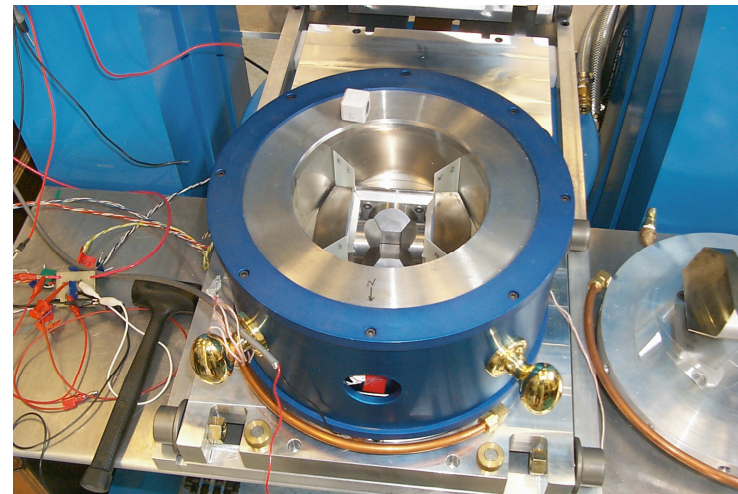
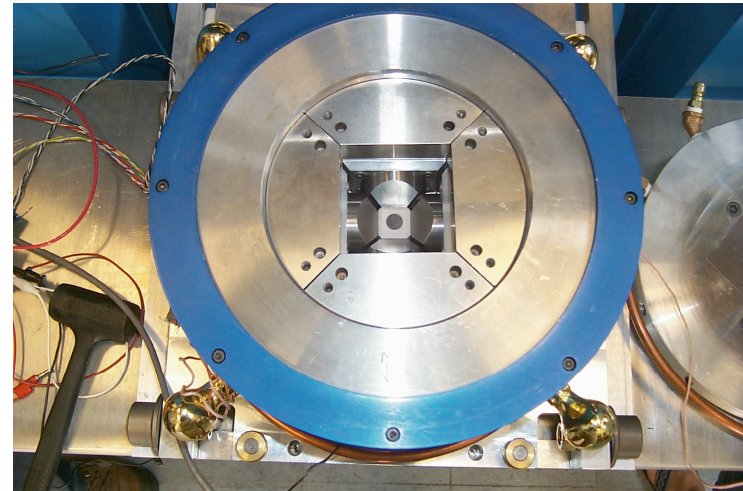
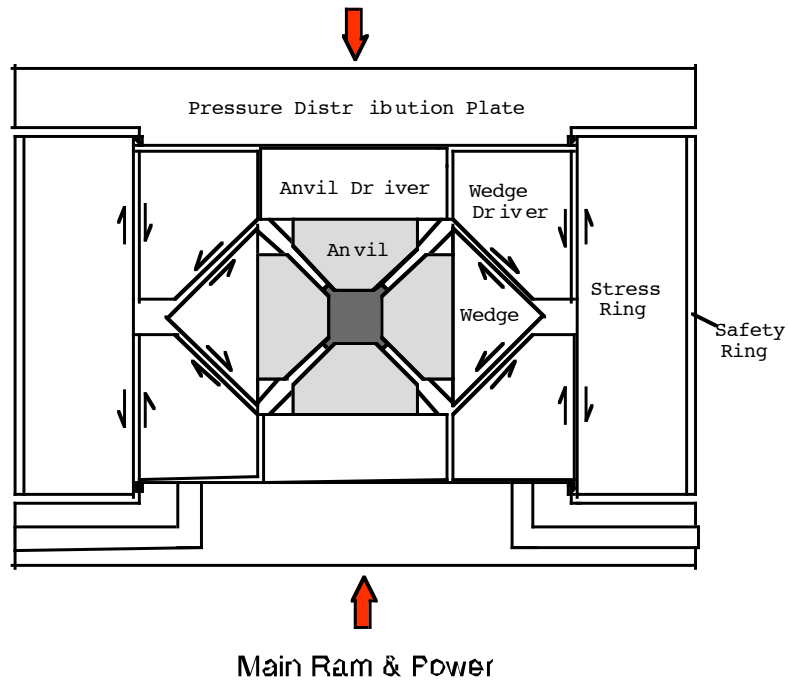
Octahedral cell

Cylindrical guide block/unanchored wedges

Walker 6-8 module

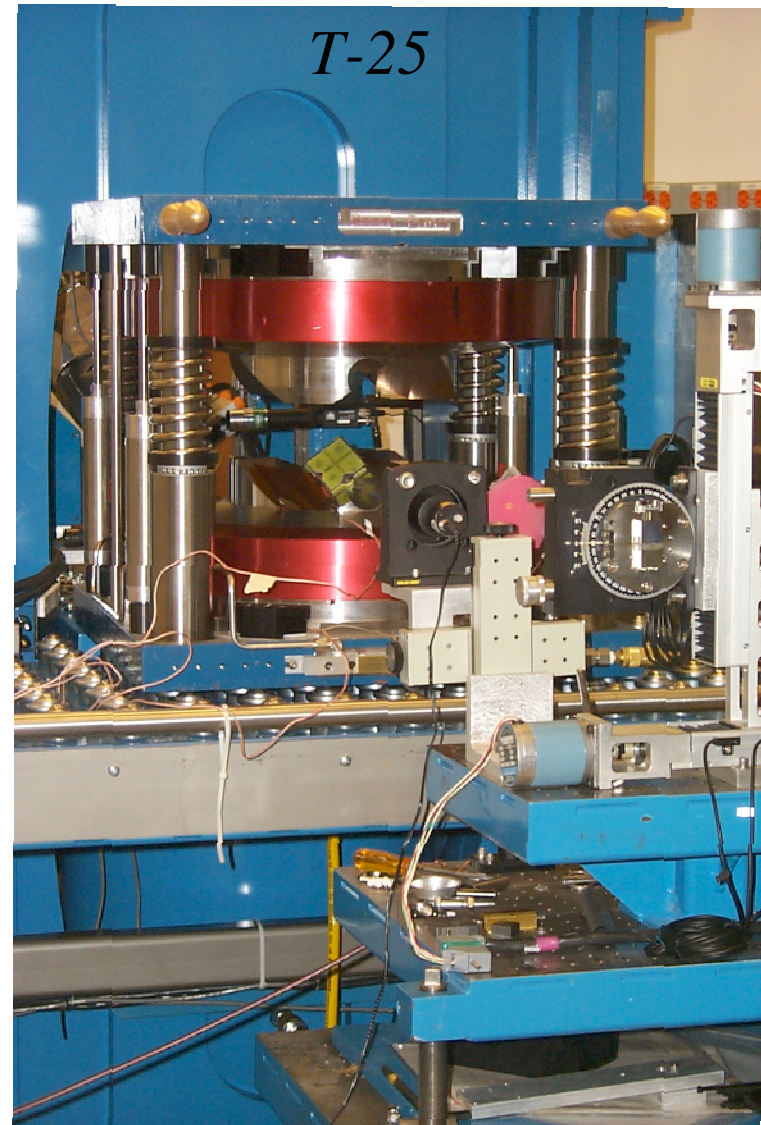
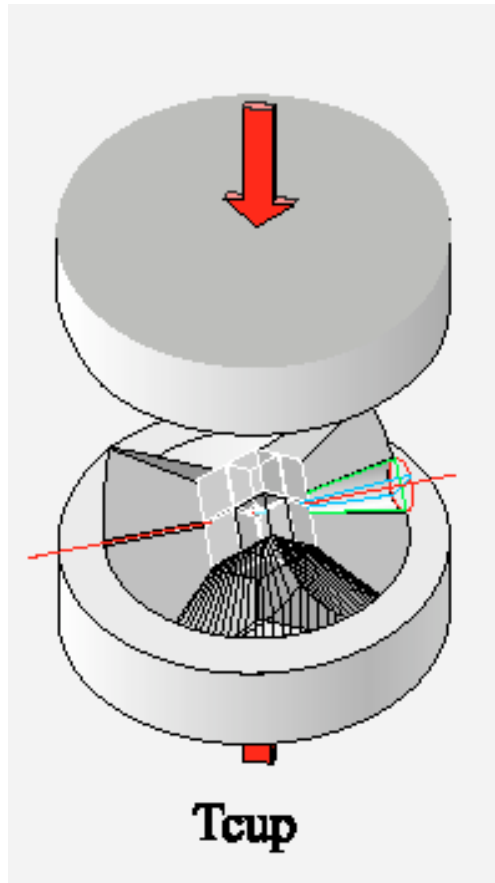


Getting 8-6 (cubic) module



24/19 assembly

Split-cylinder 6-8 systems



Bayreuth Large Volume Multianvil Apparatus

Split-cylinder

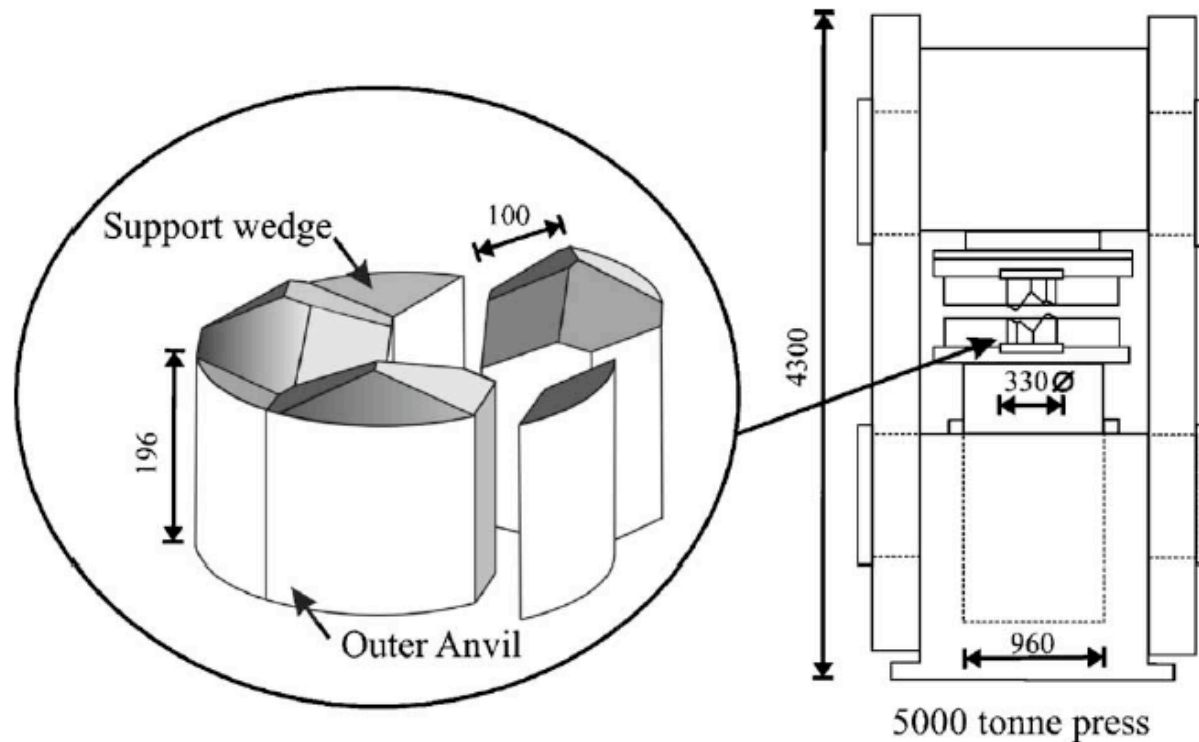
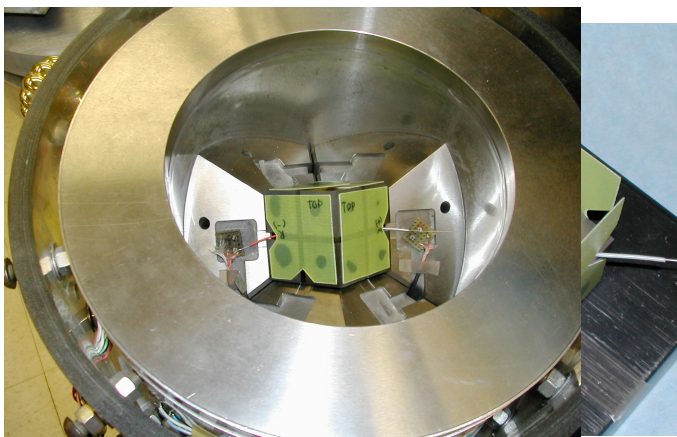
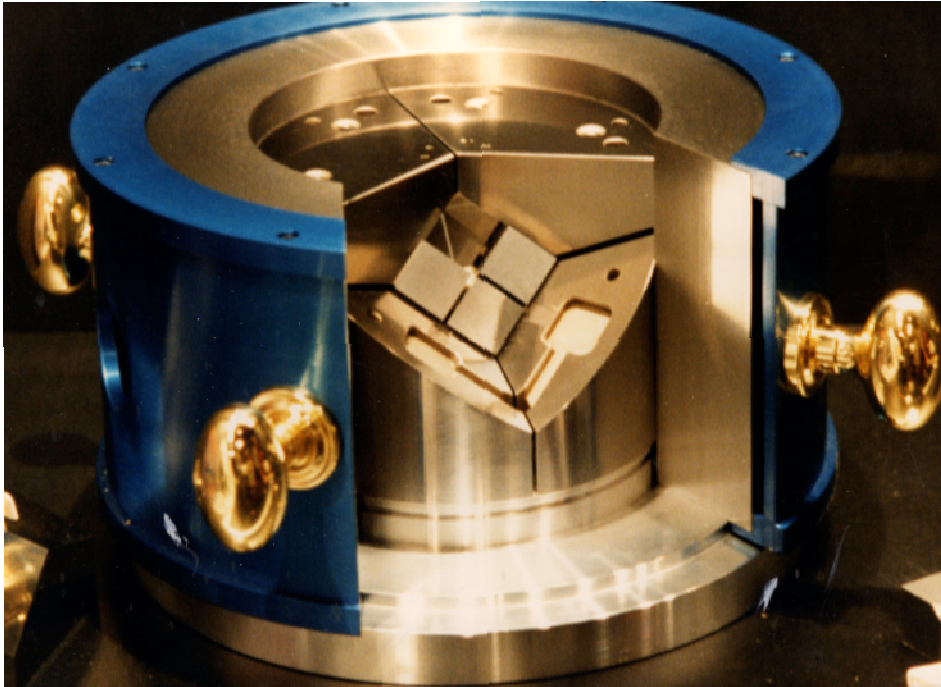


Fig. 1. Schematic diagram of the six wedges of the guide block. The three outer-anvils are separated along the outer circumference by three support wedges (see text for details). The inset to the right shows the 5000 t (50 MN) press frame with the two guide blocks inside steel platens on top of the 960 mm diameter ram. All dimensions are in millimetre.

Walker 6/8 module



Materials:

Pressure distribution plates:
aircraft-quality aluminum base,
hardened stainless steel (RC44-46)
contact plate

Stress/containment ring: M2 tool
steel, upset forging (RC38-40)

Safety ring: ductile SS w/
aluminum shield

Wedges: M2 tool steel (RC44-46)

Anvils: submicron-grade, sintered-
HIP, tungsten carbide; cubic BN;
diamond

Glide surfaces: PTFE coated mylar;
epoxy-filled fiberglass (G10) pads

Loading and Friction

Walker 6-8 module

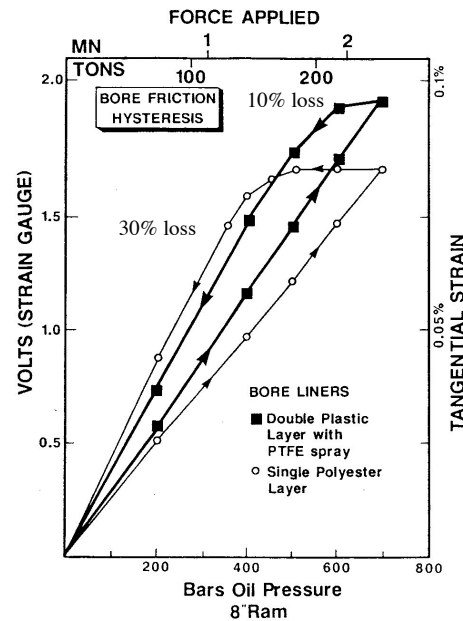
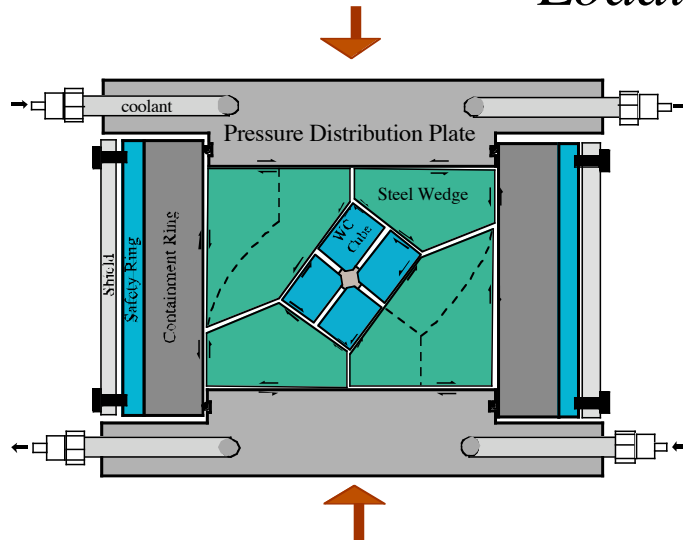
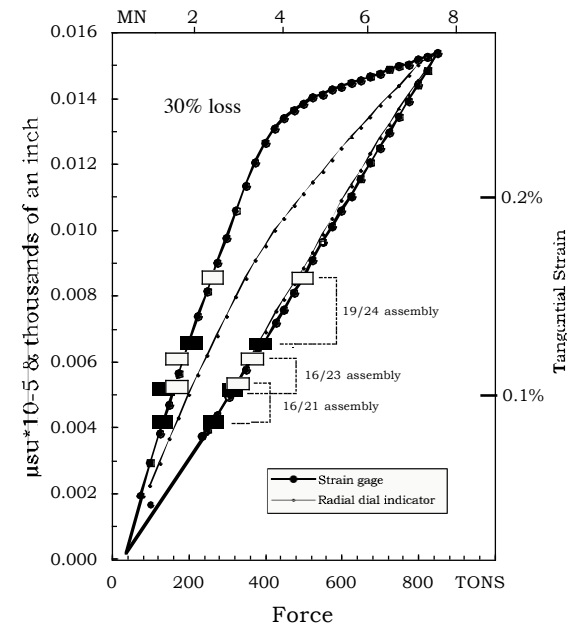
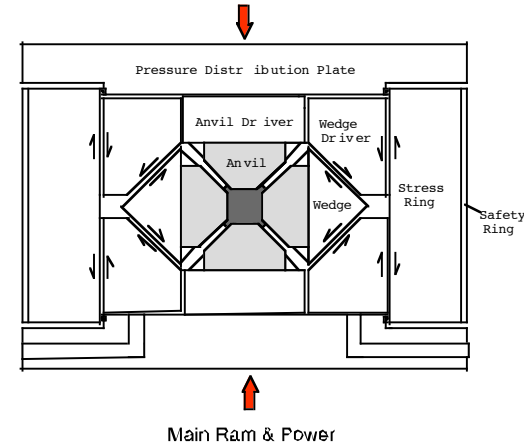


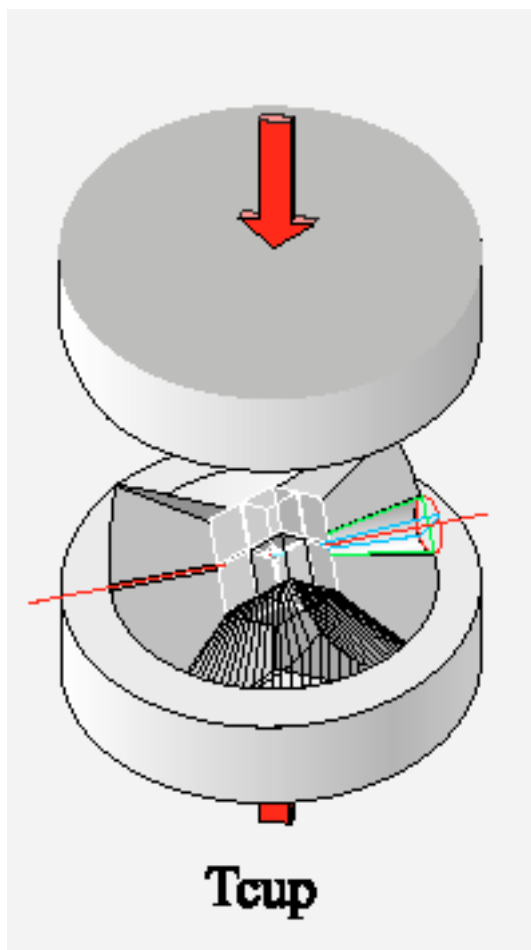
Fig. 3. Strain vs. load on press. Linear loading curves show markedly nonlinear unloading. Bore friction on the wedges is reduced through the expedient of providing a lubricated slip surface of plastic on plastic. Walker et al., 1990



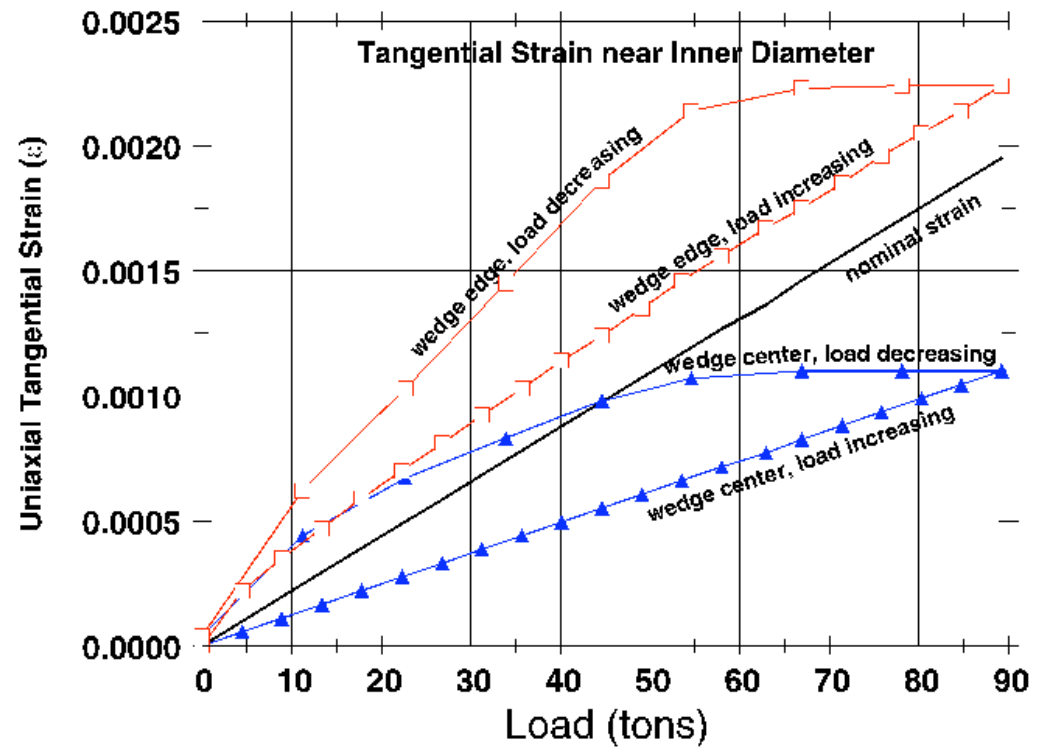
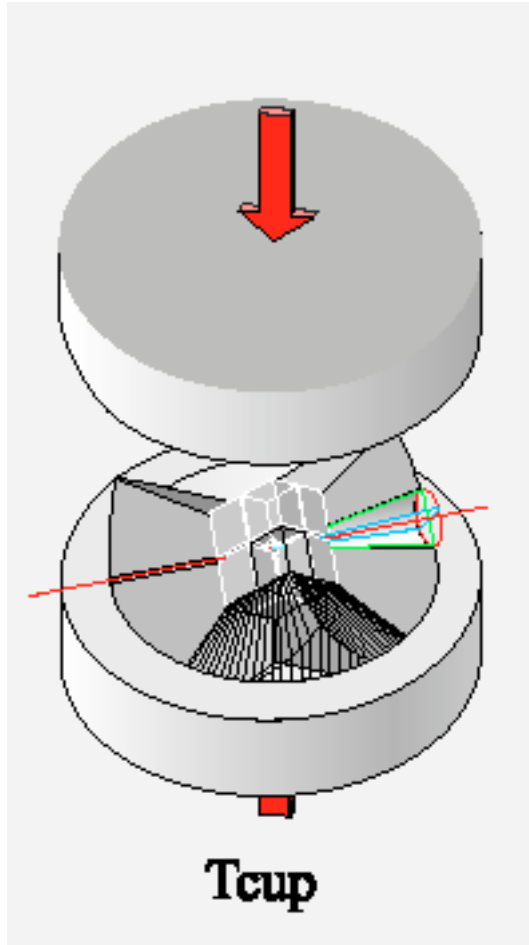
Swelling of the stress ring with the cubic anvil wedge nest and assembly under the uniaxial load from the press. The wedge driver-stress ring interface is lubricated by two sheets of mylar coated with PTFE spray. The heavy line is the strain (in micro-strain units) as measured by strain gauges. The light solid line shows the radial swelling of the stress ring (in thousands of an inch) as measured by dial indicator. The diameter of the stress ring increased by 0.030" (0.28% strain) on loading of the wedge nest to 850 tons and is fully recoverable on decompression. Non-linear unloading reflects binding of the wedges by ring. Extent of hysteresis suggest > 30% of the thrust lost to friction along the contacting surfaces of the bore and wedge drivers and wedge drivers and wedges. The hysteresis for Bi metal transitions shown by (filled rectangles - Bi I-II; unfilled rectangles - Bi II-III) for various assemblies agrees well with the stress-strain loop determined for the stress ring. (Leshar et al., 2000)

Getting 8-6 module

6-8 Tcup



6-8 T_{cup}



Strain in containment ring near the inner diameter. Maximum strain is found at the boundary between two of the first-stage anvils (wedge edges).

(Vaughan et al.)

Carbide is the weakest link!

Gettling et al., 1993

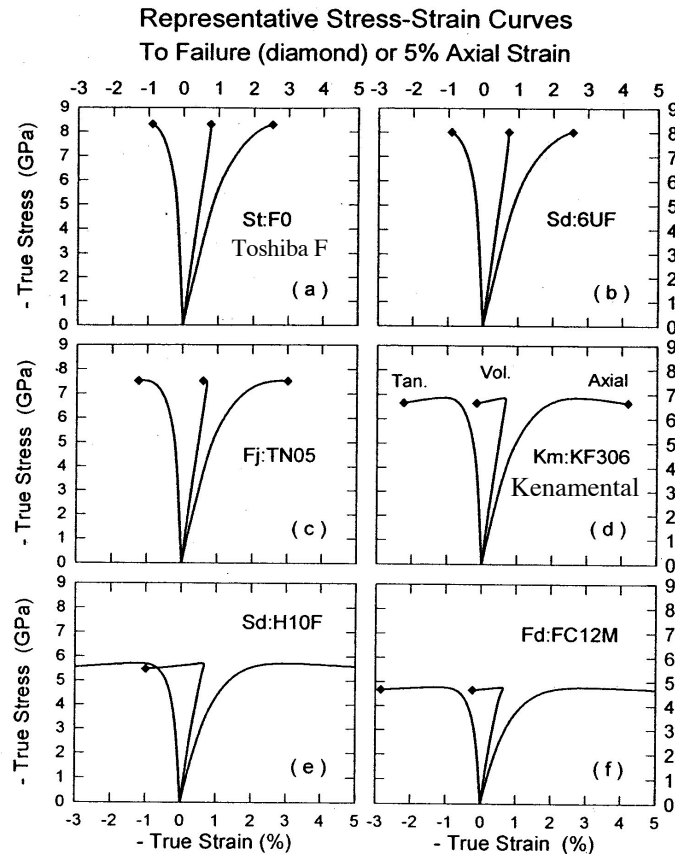
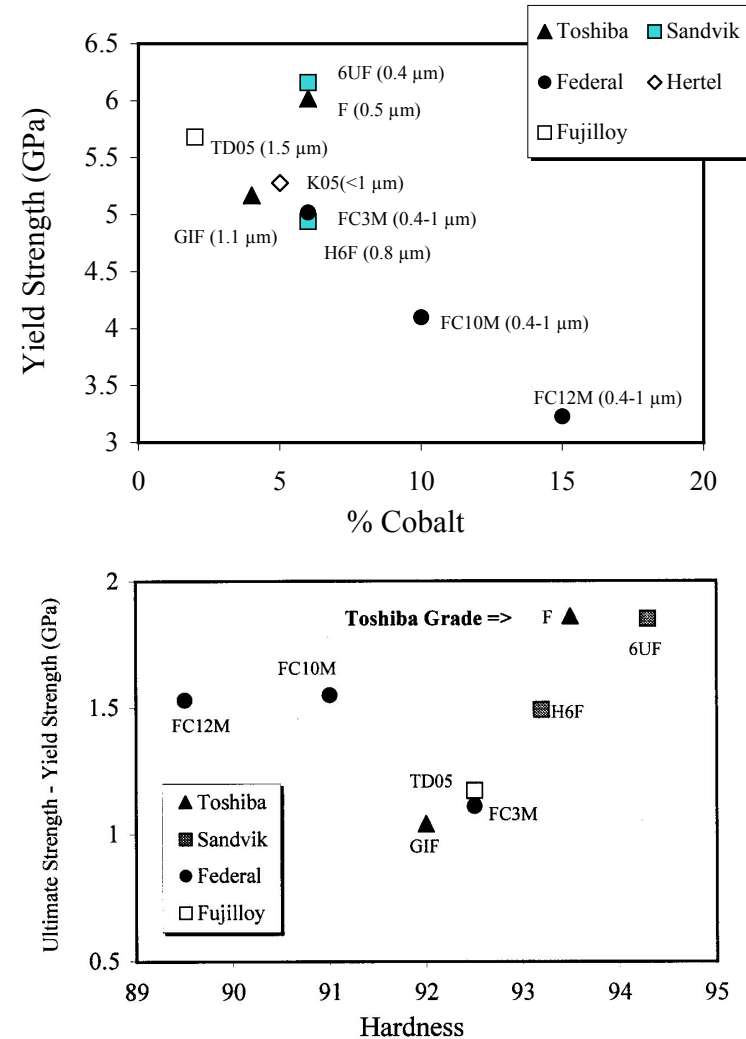


Figure 2

Representative compressive stress-strain curves for six grades. Axial, tangential, and volumetric curves are shown for each grade as identified in (d). The two strongest grades, (a) and (b), failed before reaching their ultimate potential stress as demonstrated by the positive slope of the axial stress-strain curve at failure. The other four grades achieved their maximum potential stress and all exhibit some degree of dilatancy as evidenced by the abrupt corner in their volumetric stress-strain curves and the subsequent increase in volume. Grades are designated on the plots. Manufacturer abbreviations are as follows: St: Sumitomo, Sd: Sandvik, Fj: Fujilloy, Km: Kennametal, Fd: Federal Carbide.

Submicron Grade Carbides

(Data from Gettling et al., 1993, PAGEOPH, v. 141, 545-577)



Look for carbide that is tough and hard - Toshiba F grade

Liebermann & Wang (1992).

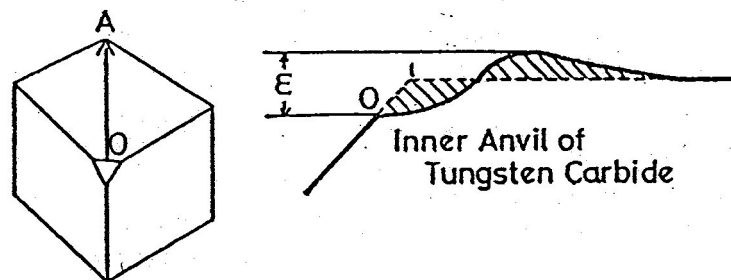


Figure 6. Typical plastic deformation observed in the tungsten carbide cubes. For the 7/2 assembly and Type C gaskets (Fig. 7, the amplitude ϵ of the deformation after achieving the GaAs transition has been measured for the Kennametal KZ 313 (150 micron), Hertel KF1 HIP (100 micron) and Toshiba F (75 micron) grades.

Gwanmesia et al. (1993)

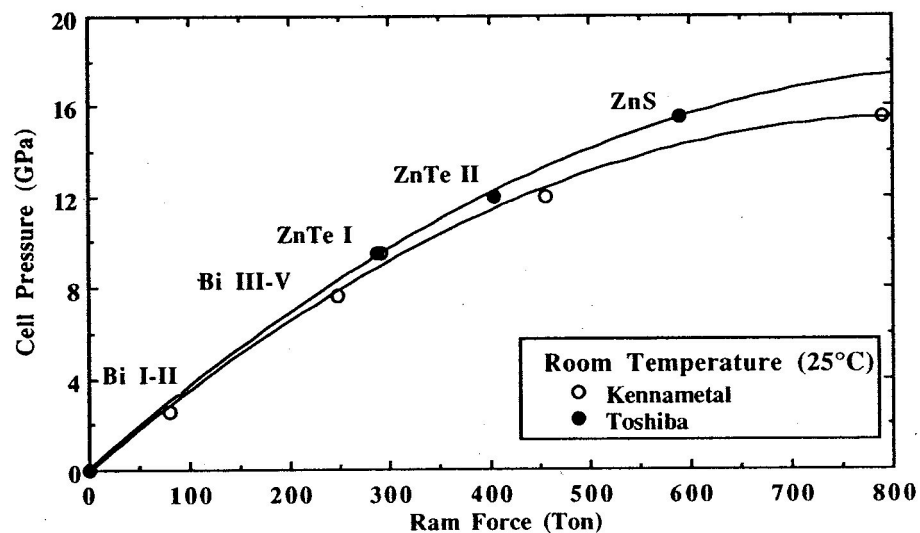
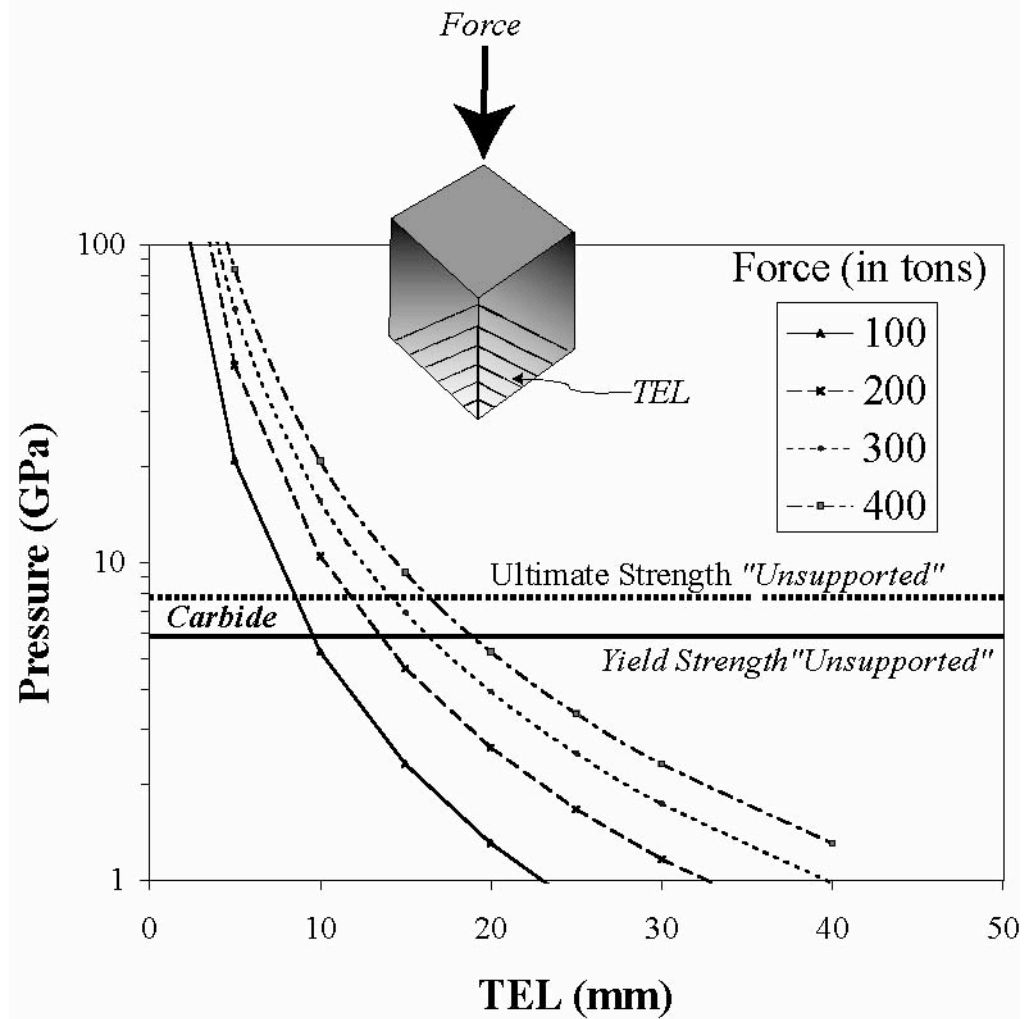
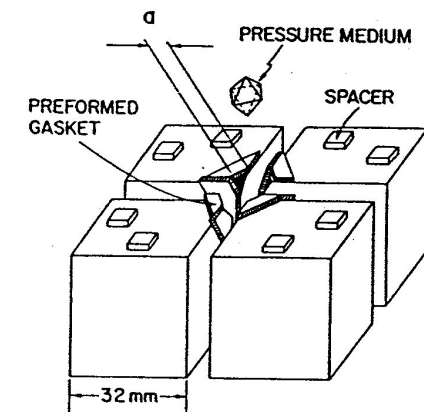
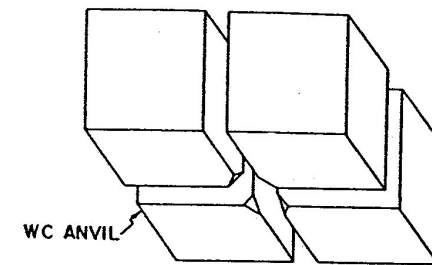
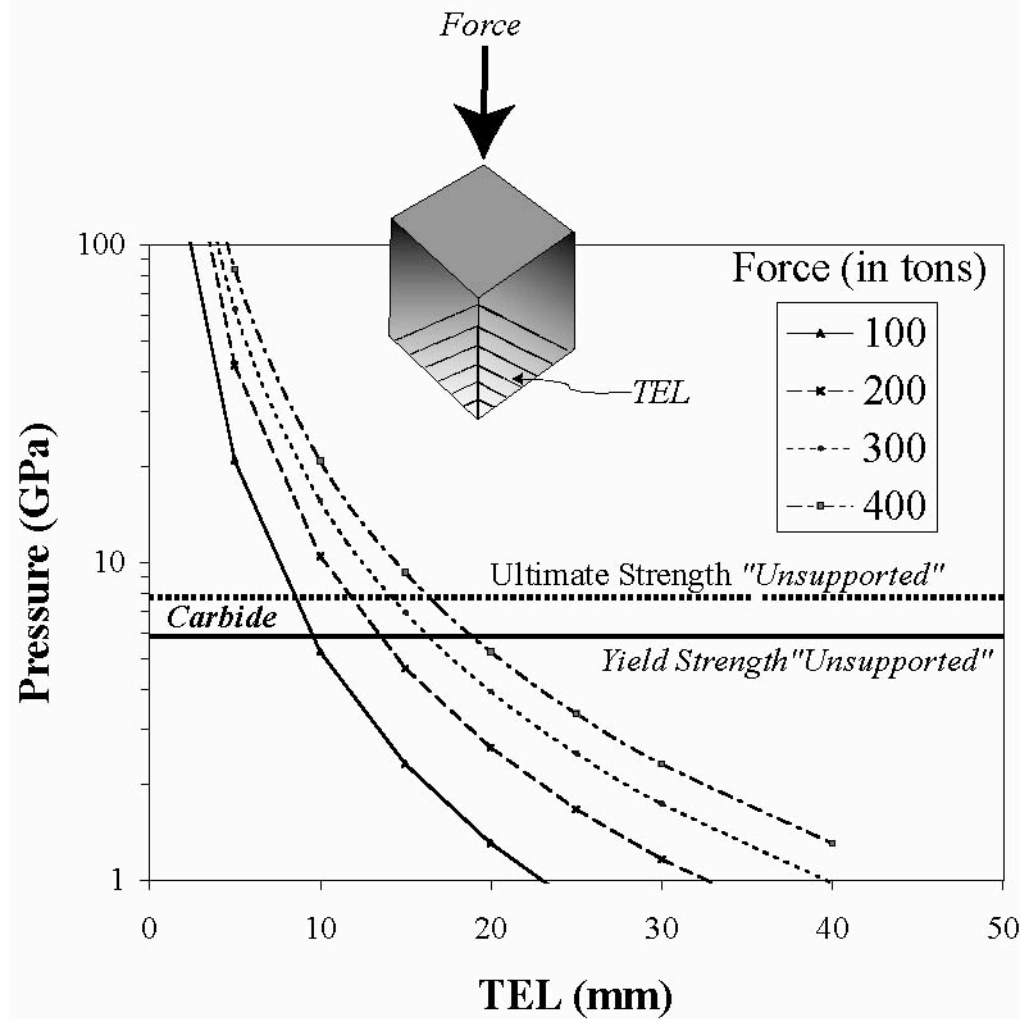


Figure 6
Pressure calibration of 14/7/5 cell assemblies at room temperature using both Kennametal (open circles) and Toshiba (closed circles) tungsten carbide cubes.

Importance of gasketing



Importance of gasketing



Cell Assemblies

Octahedron

pyrophyllite

semi-sintered MgO (\pm Cr₂O₃; machined or casted)

Gaskets

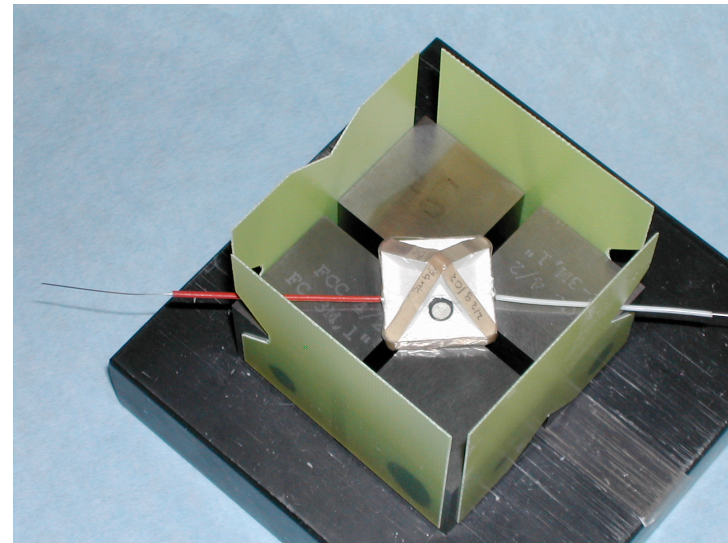
pyrophyllite (preformed)

semi-sintered MgO (cast w/ octahedron)

-back-up Teflon tape-

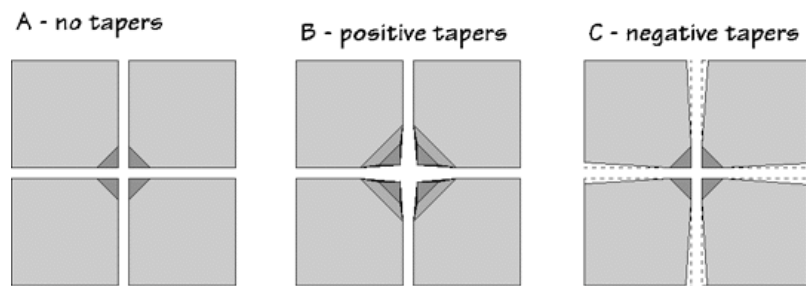


COMPRES assembly
(K Leineweber)



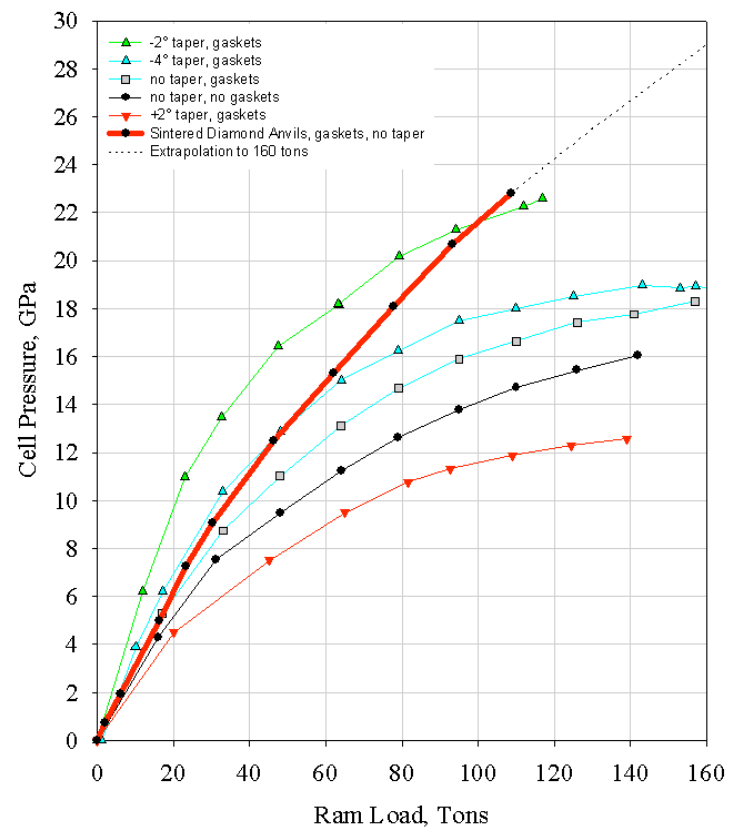
Material: Aremco 584 (MgO)
Porosity: 40% (fired)

<http://www2.bnl.gov/sam85/tcup/html/tcup.html> (Vaughan et al.)



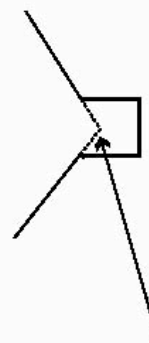
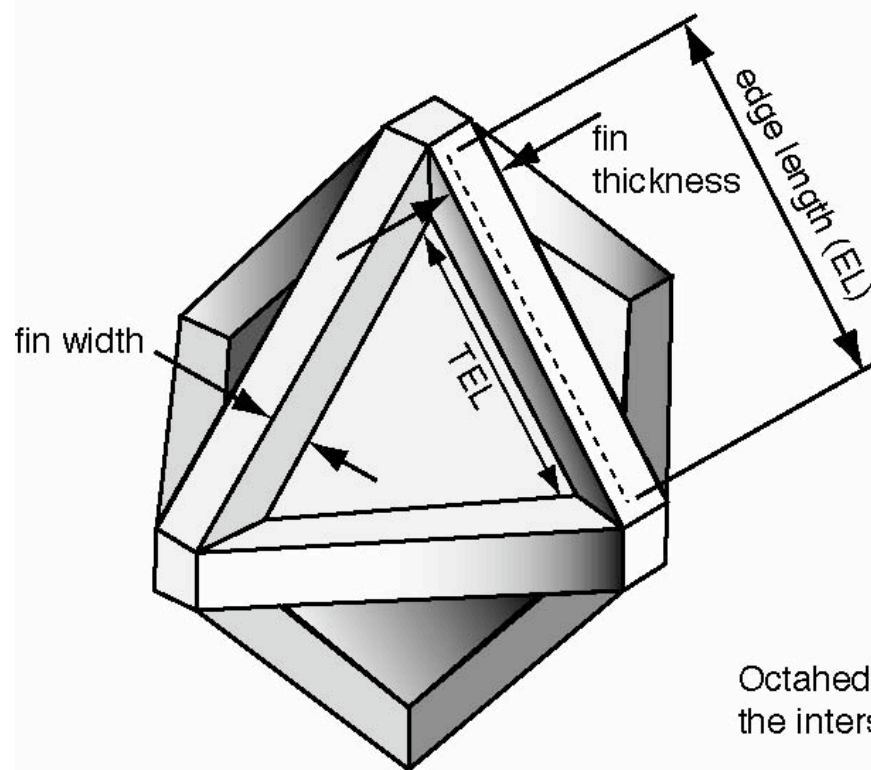
Effect of Tapering and Gaskets

7mm octahedra, 2mm truncations

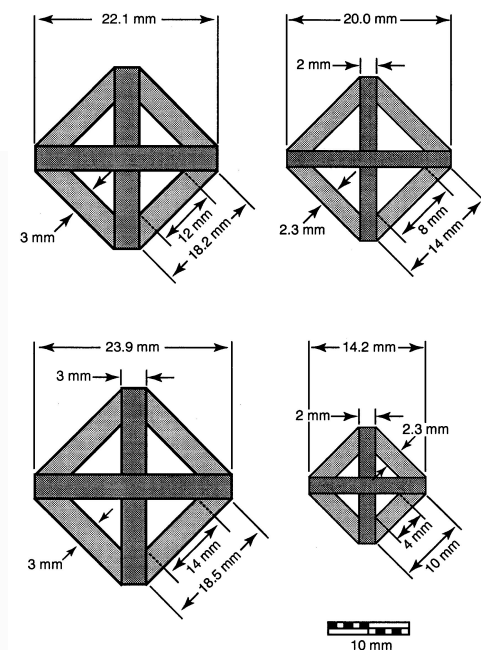


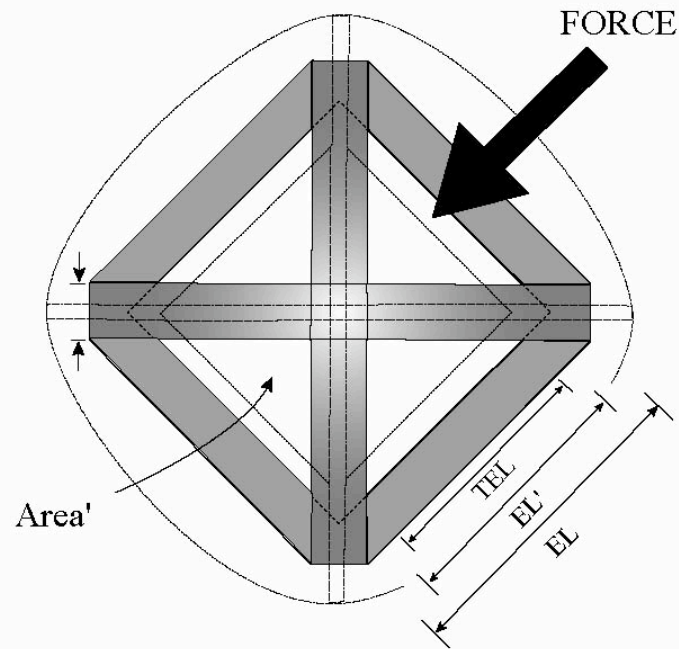
The Pressure Cell

Perspective view of Castable MgO octahedra



Octahedra 'edge length' is measured from the intersection of the faces of the octahedra.





$$V_{\text{original}} = \sqrt{2/3} (EL)^3$$

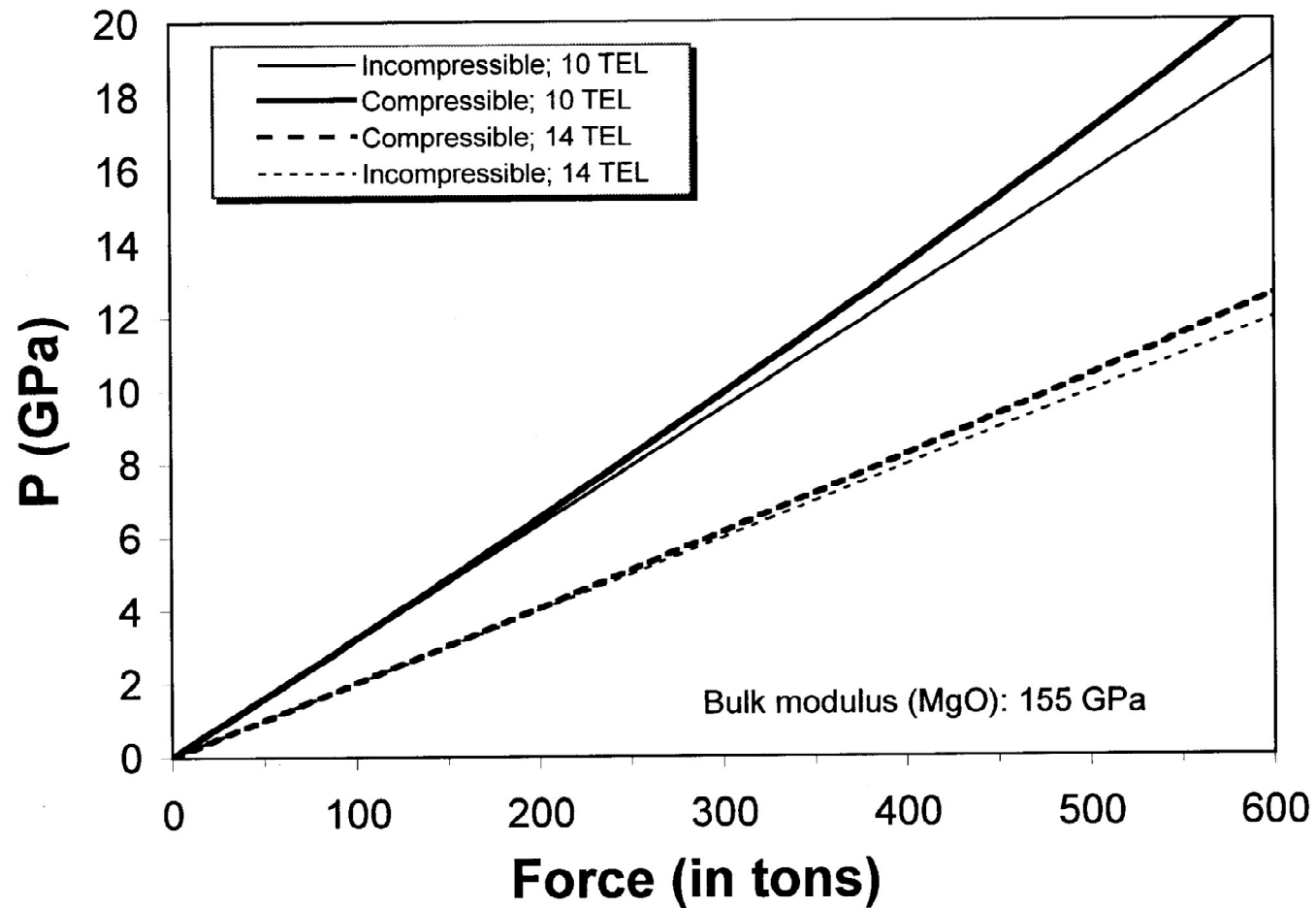
Porosity correction*: $V_{\text{densified}} = V_{\text{original}} (1 - \phi_{\text{fired}}) = \sqrt{2/3} (EL')^3$

$$P = 0.25 \text{ Force/Area}' = 0.5 \text{ Force/(EL' \cdot height)}$$

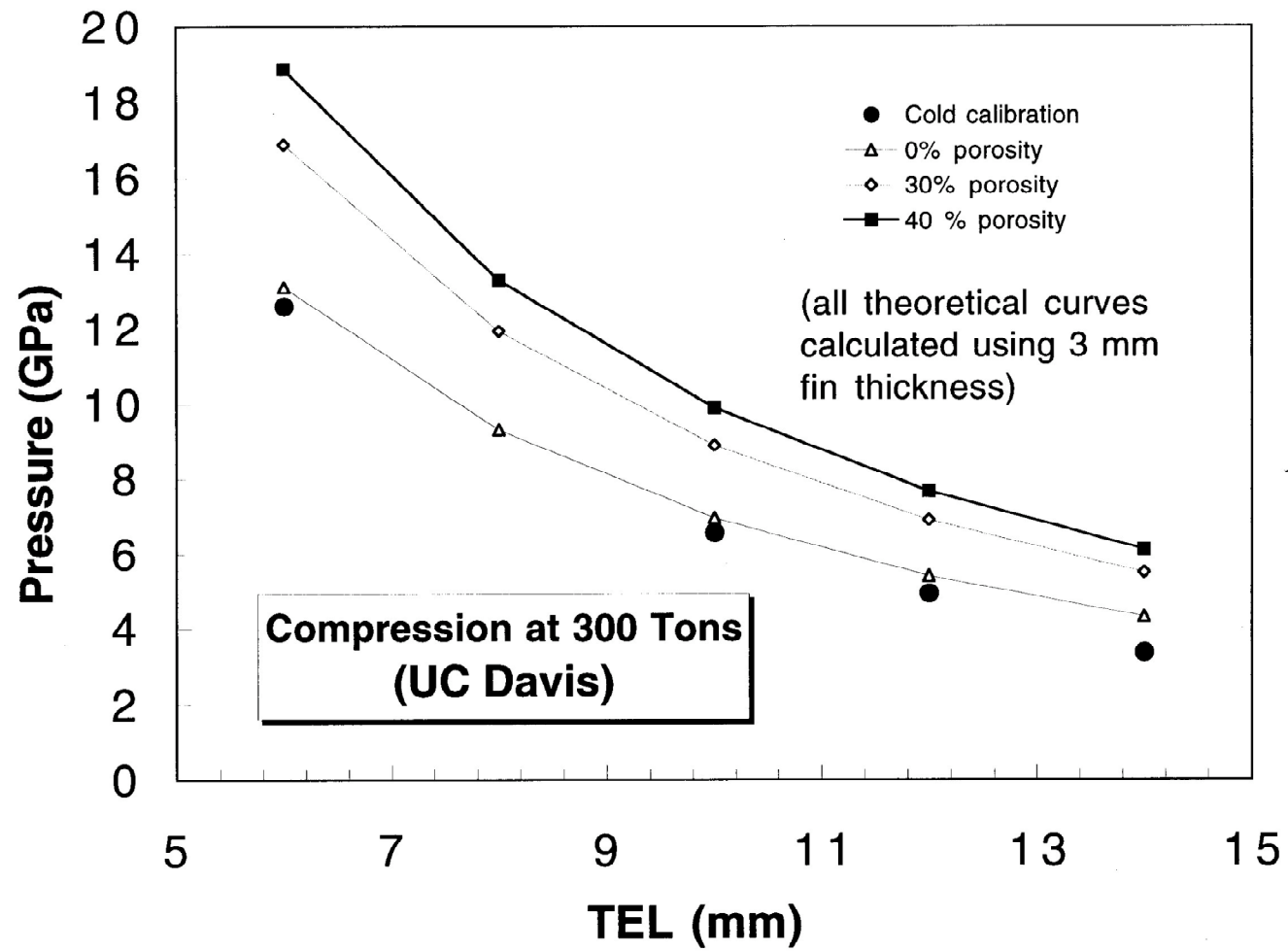
Compressible medium: $P \sim \Delta V/V_0 K$

*Aremco 584 (MgO): 40% porosity (fired)

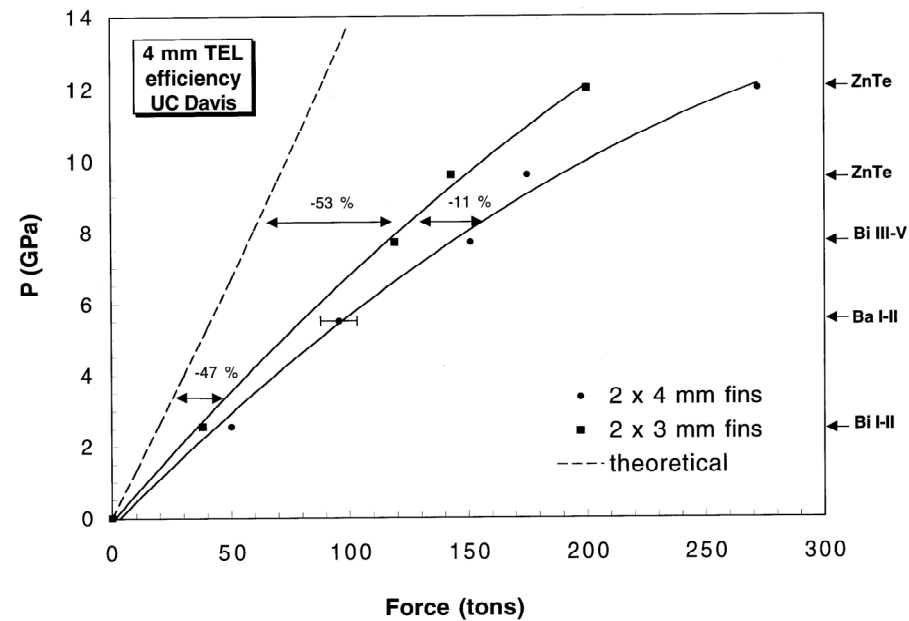
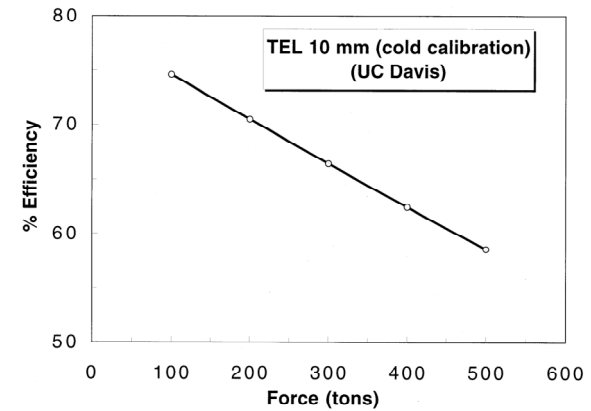
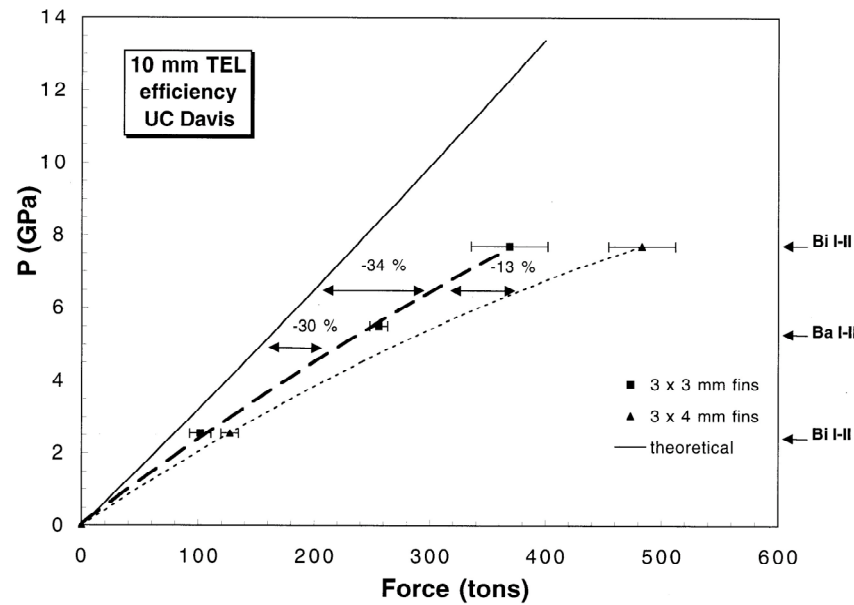
Incompressible vs. compressible



Porosity



Carbide support vs. Efficiency

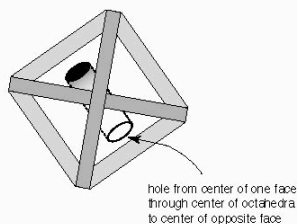


Heat Generation

Furnace Types

Graphite

- straight-wall or stepped
- resistor; 20:1 stepdown
- < diamond stability

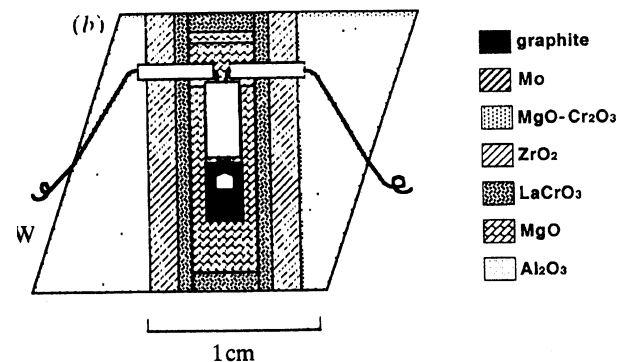


LaCrO₃

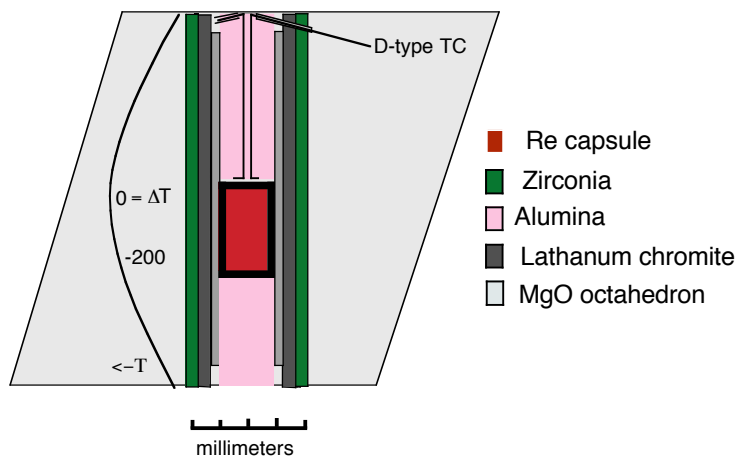
- straight-wall or stepped
- semiconductor at h-T; <10:1 stepdown

Re foil, Inconel

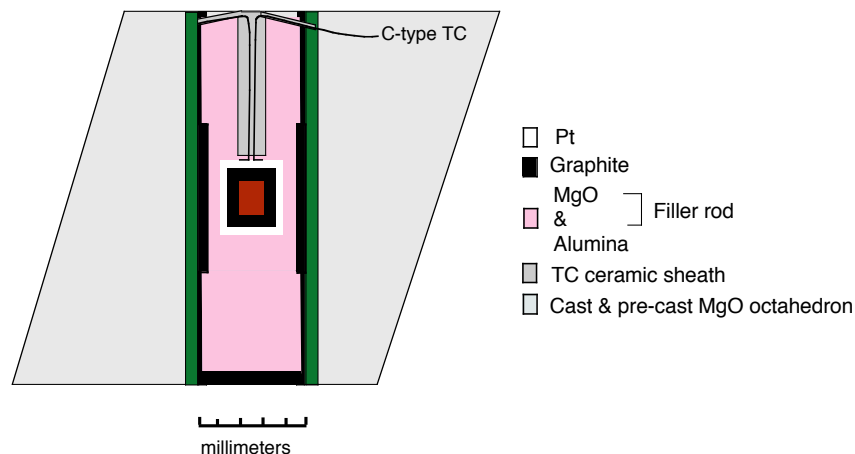
18/11 Assembly



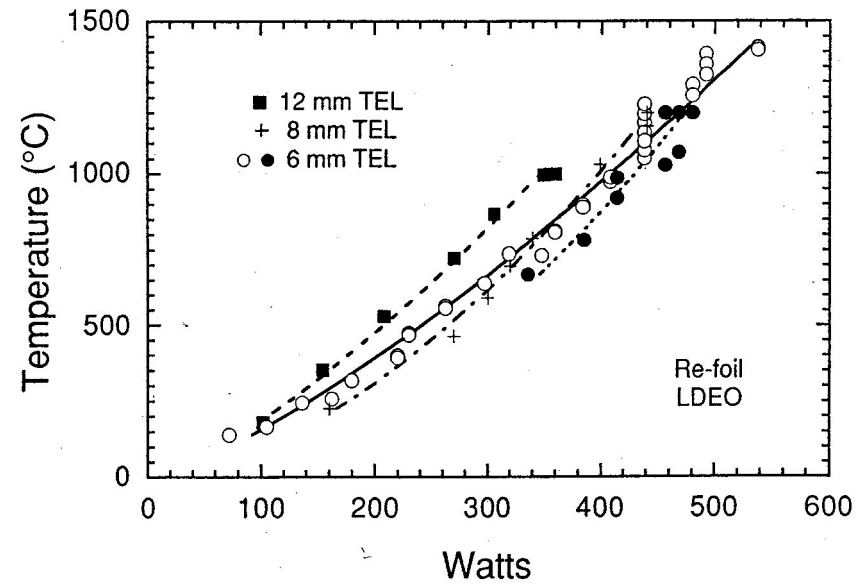
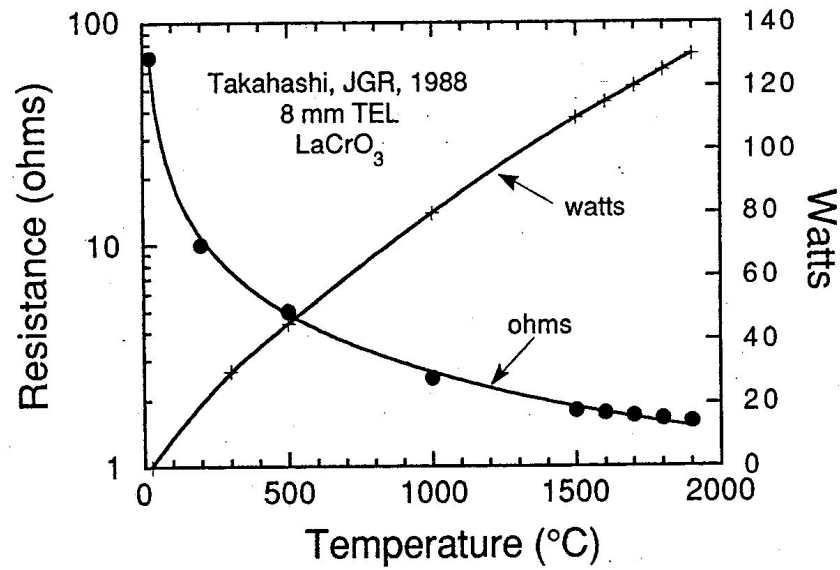
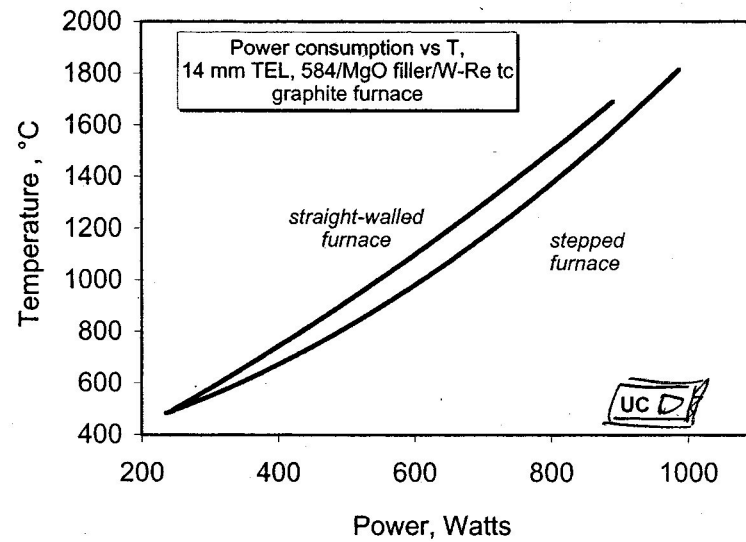
14/8 Assembly



18/11 Assembly



Power Consumption



Temperature Control

Thermocouple Configurations

Radial

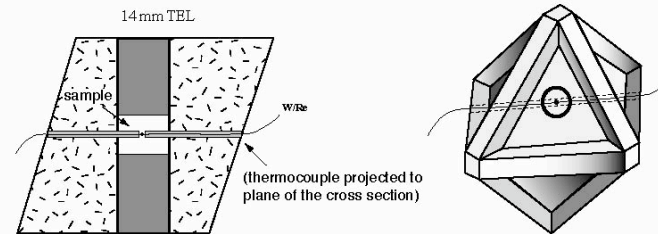


Figure 1. Cross-section and perspective view of assembly with thermocouple wire passed directly through sample in center of assembly.

Axial

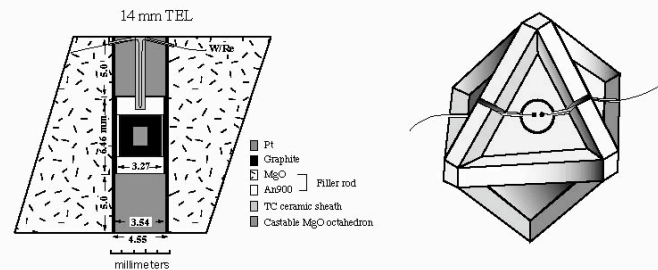
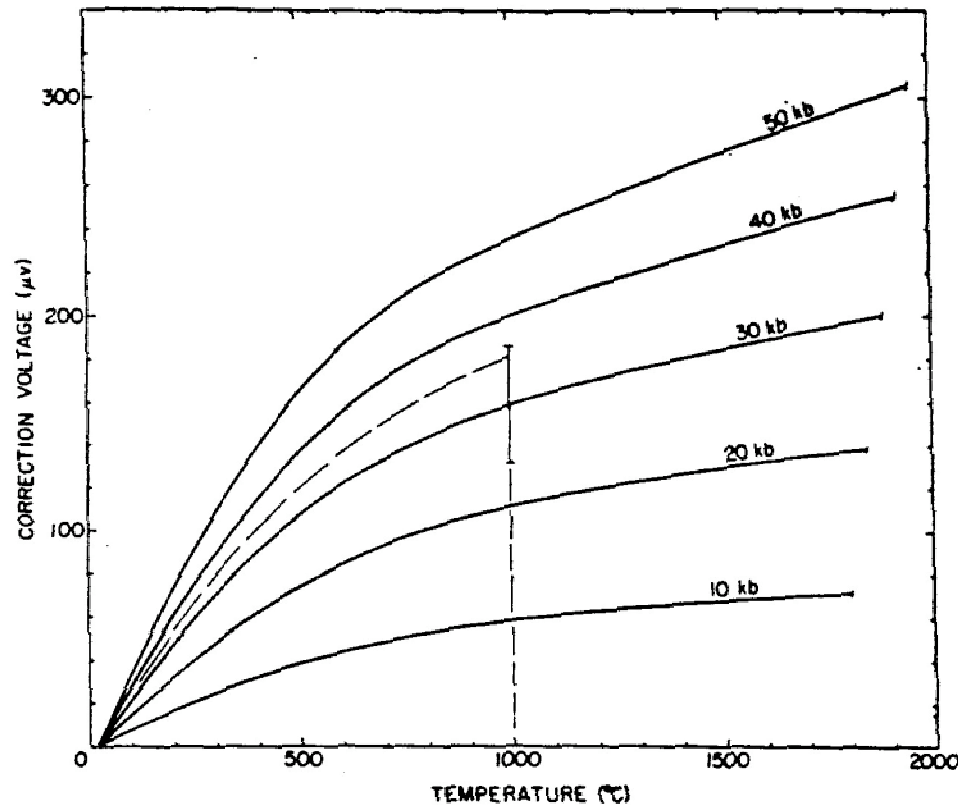


Figure 2. Cross-section and perspective view of assembly with thermocouple wire brought down through top of furnace. Wires are passed through channels cut into octahedra that are filled with alumina cement.

P/T dependence of emf



S-type TC

FIG. 11. Corrections for Pt-Pt10Rh thermocouples operating with a seal temperature of 20°C. The dashed lines indicate the upper limits of the experimentally explored region.

Gettings & Kennedy (1970)

Poorly characterized for W/Re thermocouples

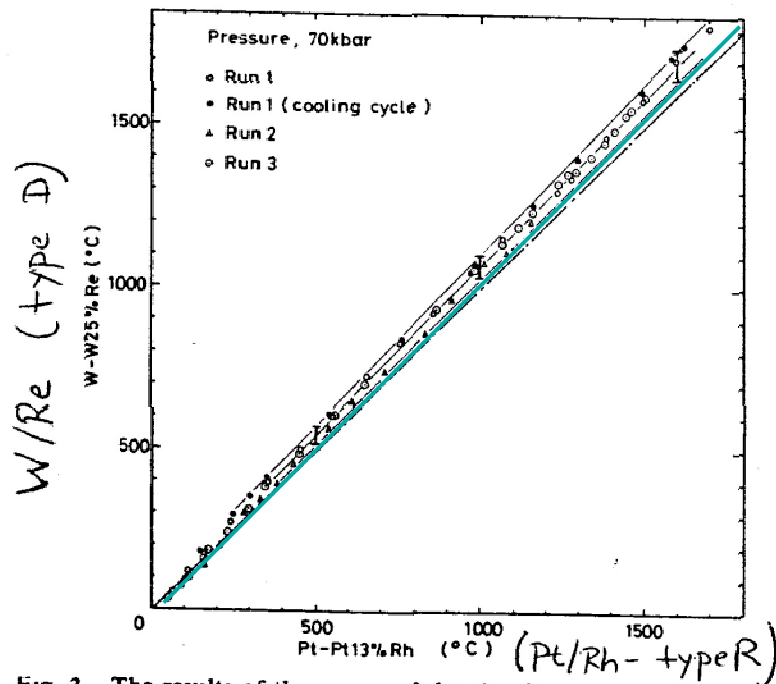
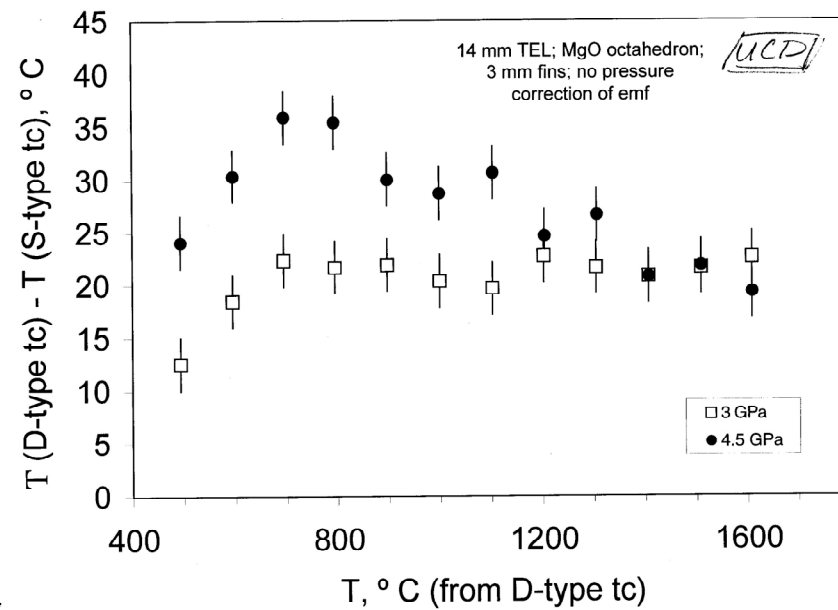


Fig. 3. The results of three runs of the simultaneous measurements of the temperature values indicated by Pt-Pt13%Rh and W-W25%Re thermocouples at 70 kbar. The solid circles are the result of the cooling cycle of Run 1. The others are the results of the heating cycle of three runs.

Ohtani (1979)



Thermal Gradients

CELL-ASSEMBLY FOR 10MM OCTAHEDRON

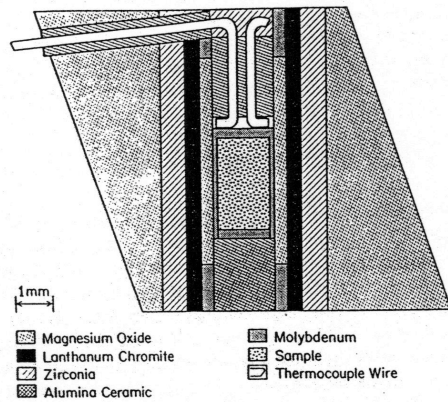


Figure 10. Cell assembly for determining the temperature distribution using the enstatite content of diopside pyroxene coexisting with enstatitic pyroxene (from Gasparik, 1989;—rhenium replaces molybdenum as the capsule in the higher temperature experiments of Presnall and Gasparik, 1990).

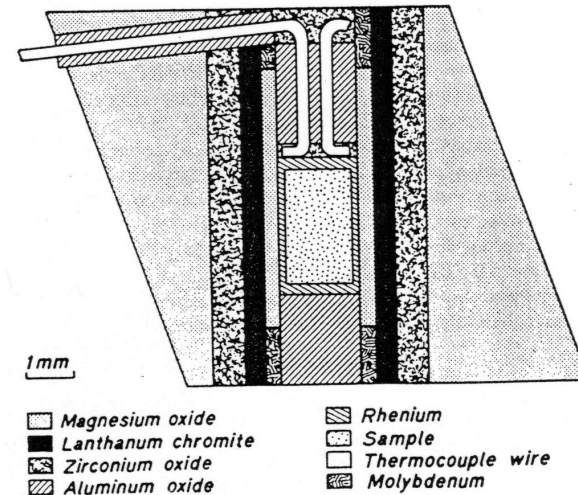


Fig. 1. Assembly of 10-mm octahedron.

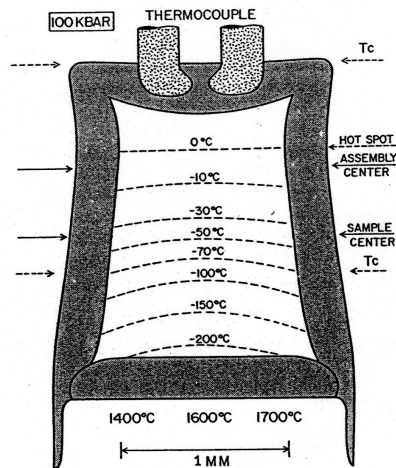


Figure 11. Cross-section of an experimental charge in a 10 mm MgO cell assembly showing the isotherms inferred from the enstatite-diopside data which are in Fig. 3 of Gasparik (1989).

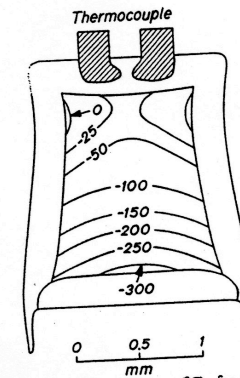
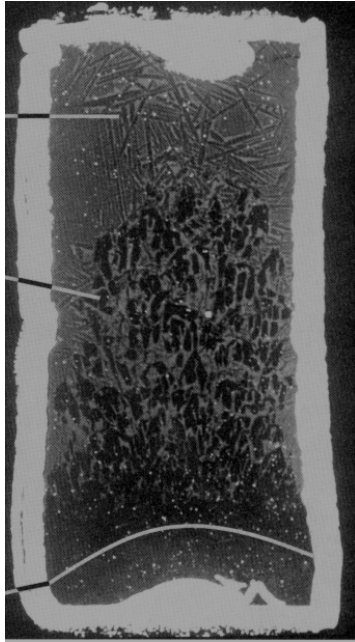


Fig. 2. Relative temperature contours ($^{\circ}\text{C}$) for experiments in the temperature range $2000^{\circ}\text{--}2150^{\circ}\text{C}$. Contours are interpolations based on Figure 3 of Herzberg *et al.* [this issue] with slight modifications of shape based on crystal-liquid isothermal surfaces observed in this study. The pressure cell assembly of Herzberg *et al.* [this issue] is identical to the pressure cell assembly used here except for their use of Mo capsules. The temperature difference of 300°C between the hot and cold ends of the capsule is consistent with the smaller difference of 200°C reported by Gasparik [1989] at lower temperatures ($1400^{\circ}\text{--}1700^{\circ}\text{C}$).

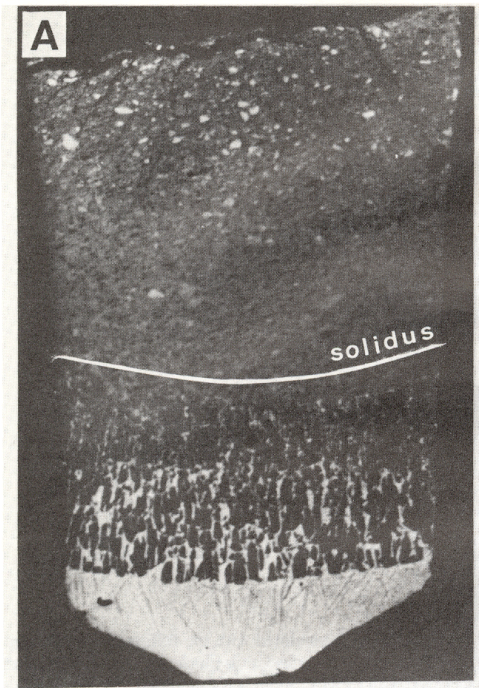
5 GPa/1950 °C/8 min



Zhang and Herzberg (1994)

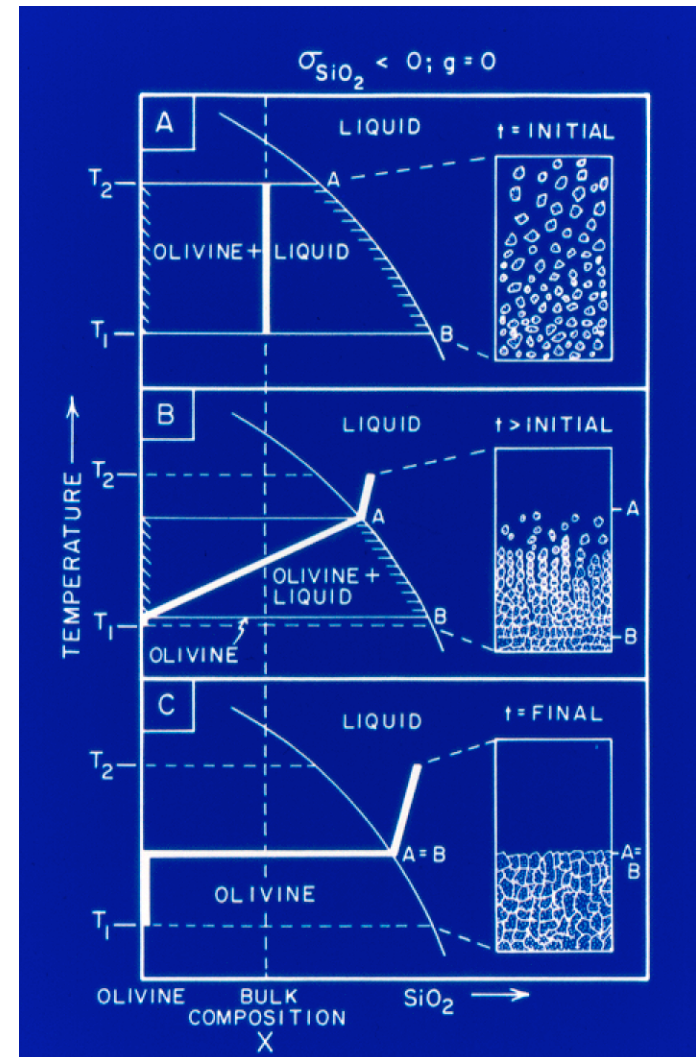
Consequences of
large thermal gradients

5 GPa/1700°C/10 min



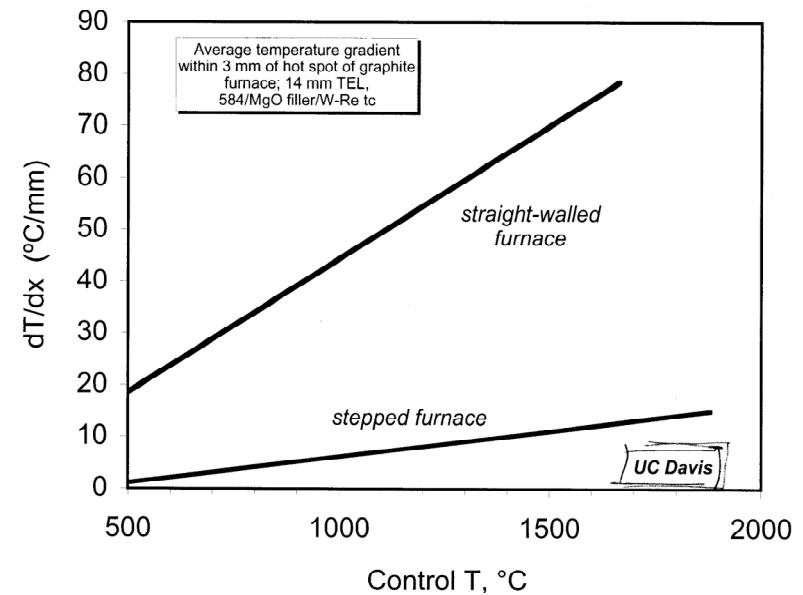
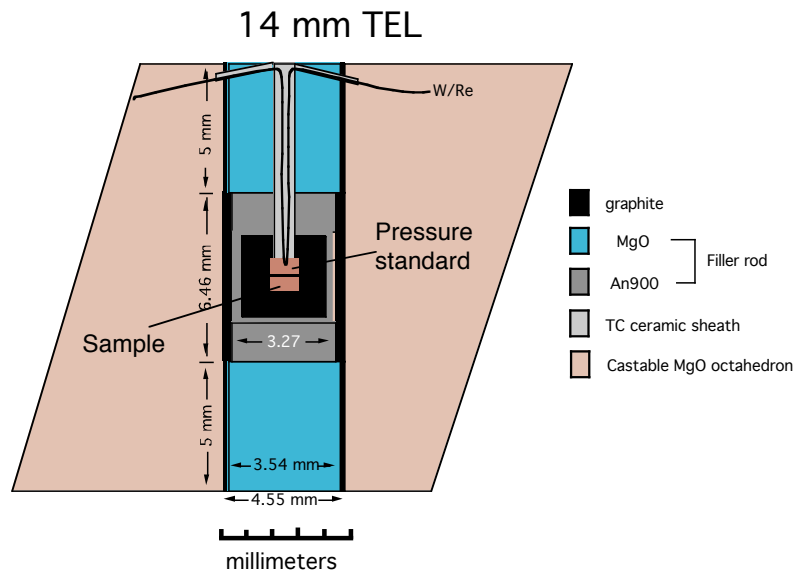
Takahashi (1986)

Thermal Migration

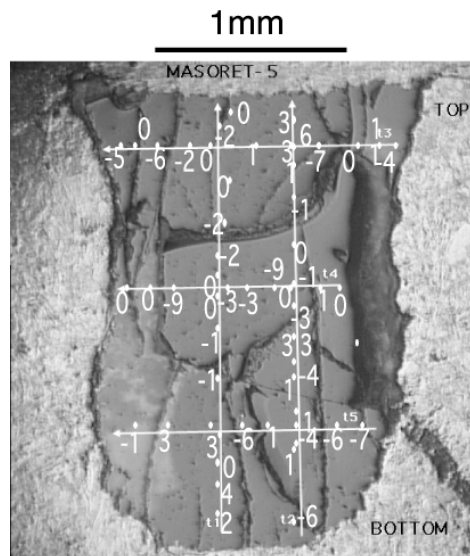


Leshner and Walker (1988)

Minimize thermal gradients

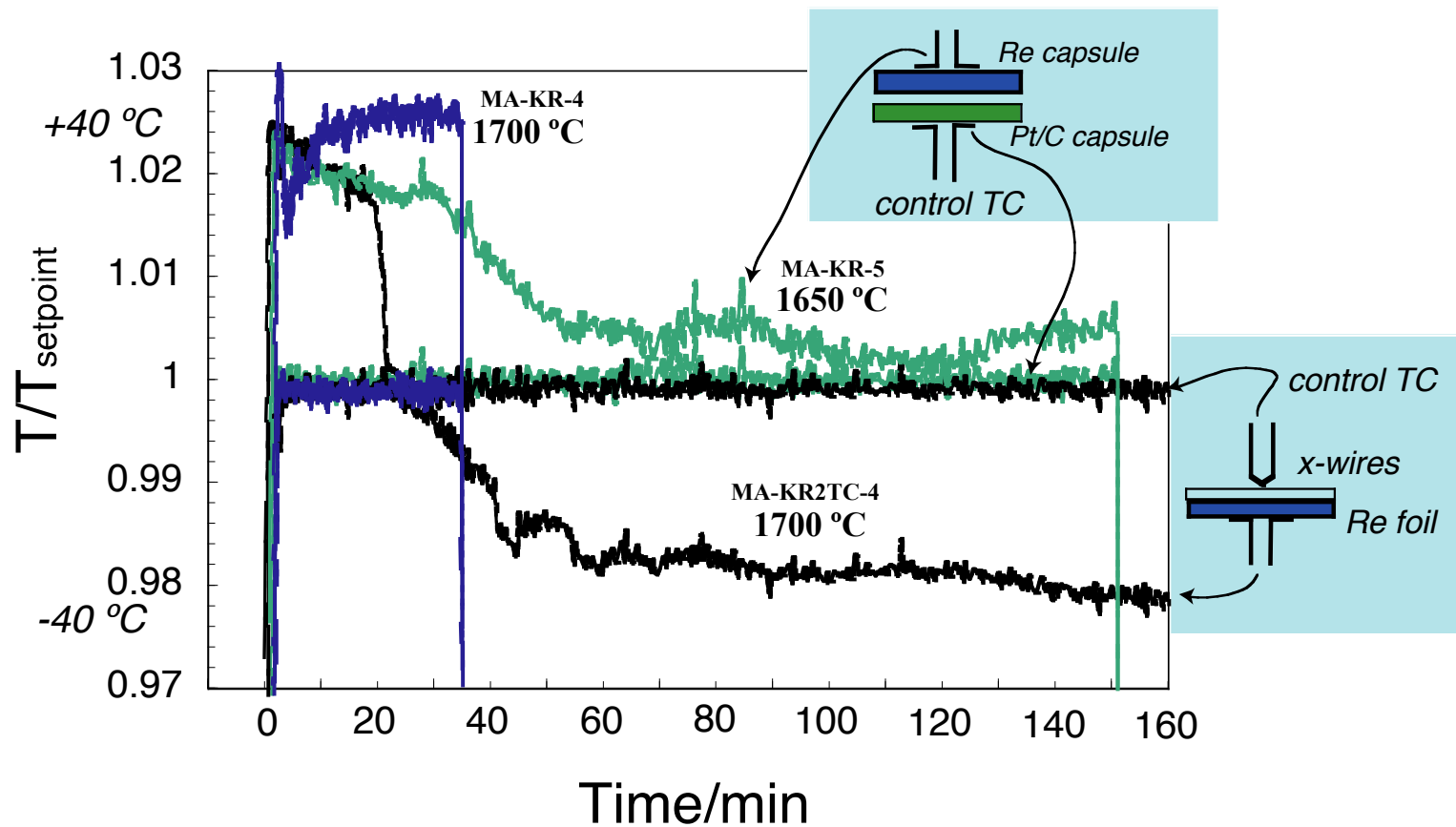


Cross-section of stepped heater assembly.



Soret Effect Calibration
($\pm 10^{\circ}\text{C}$)

Thermocouple instability/drift



- Short-time-scale *instability* - stress relief, recrystallization
- Long-time-scale *drift* - volume diffusion, contamination, oxidation

Pressure Calibration

Metal Transitions

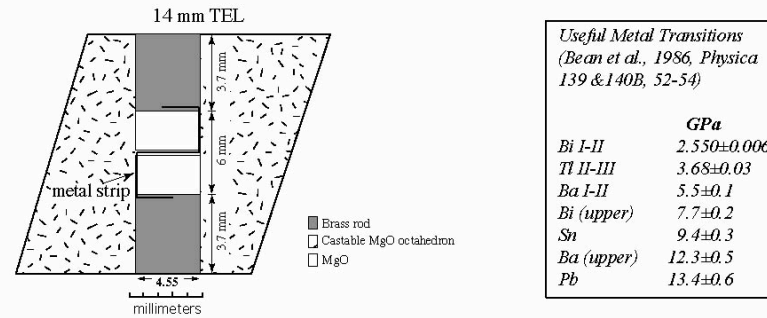
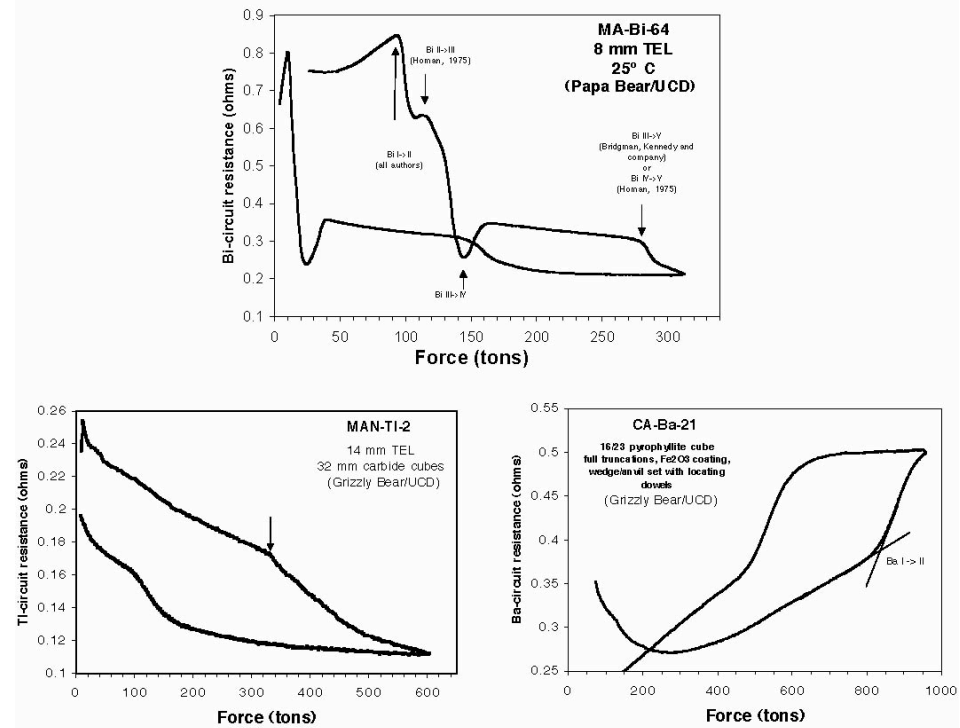
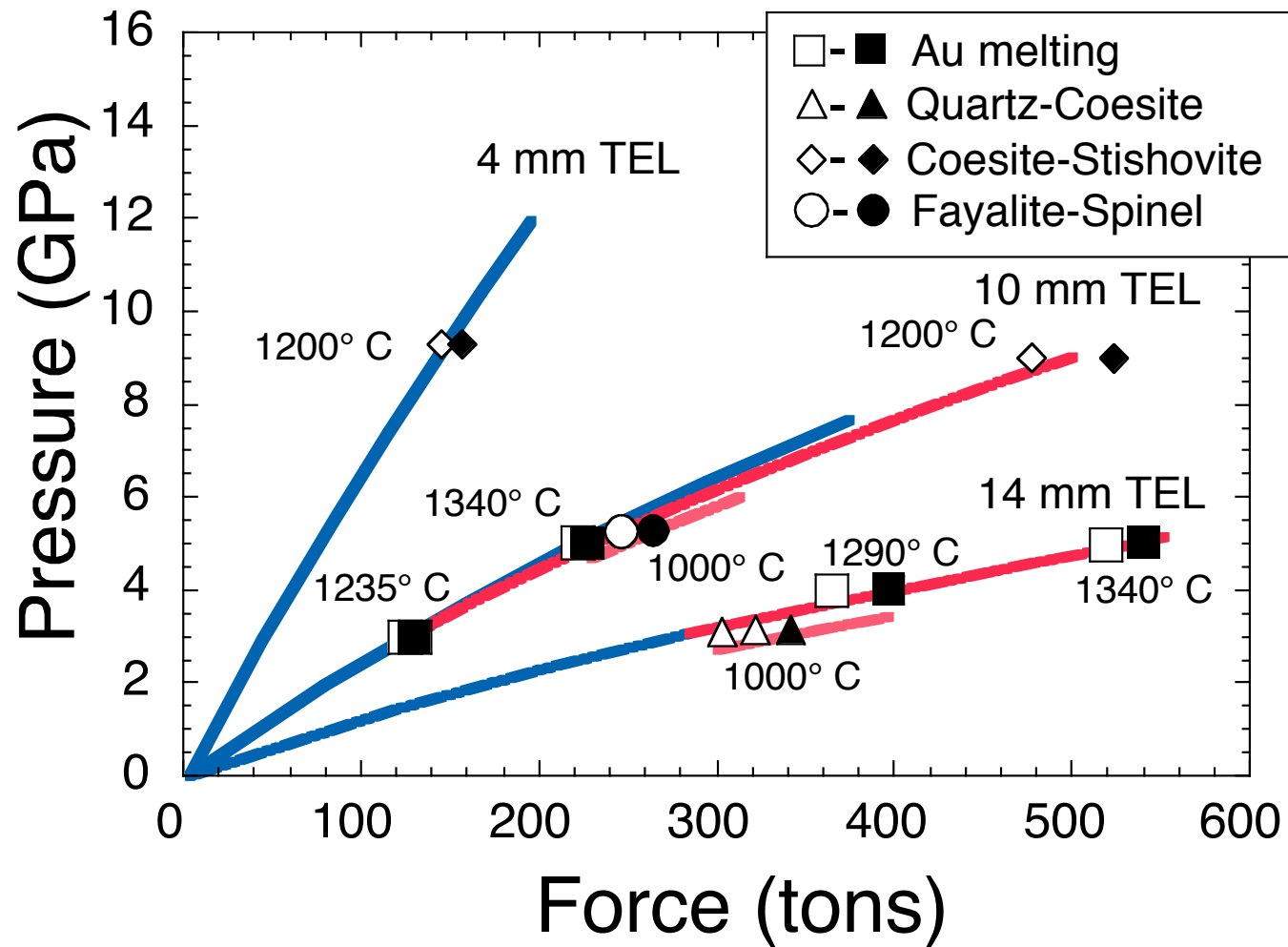


Figure 3. Cross-section of a metal transition assembly.



Phase Bracketing



Phase Bracketing

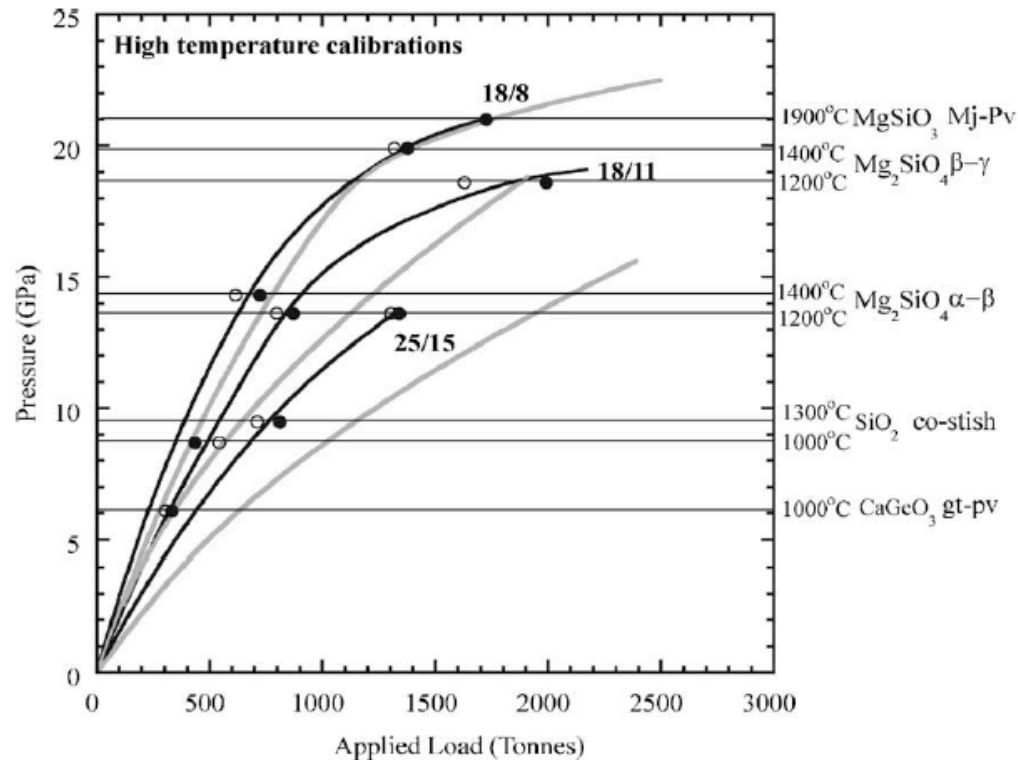
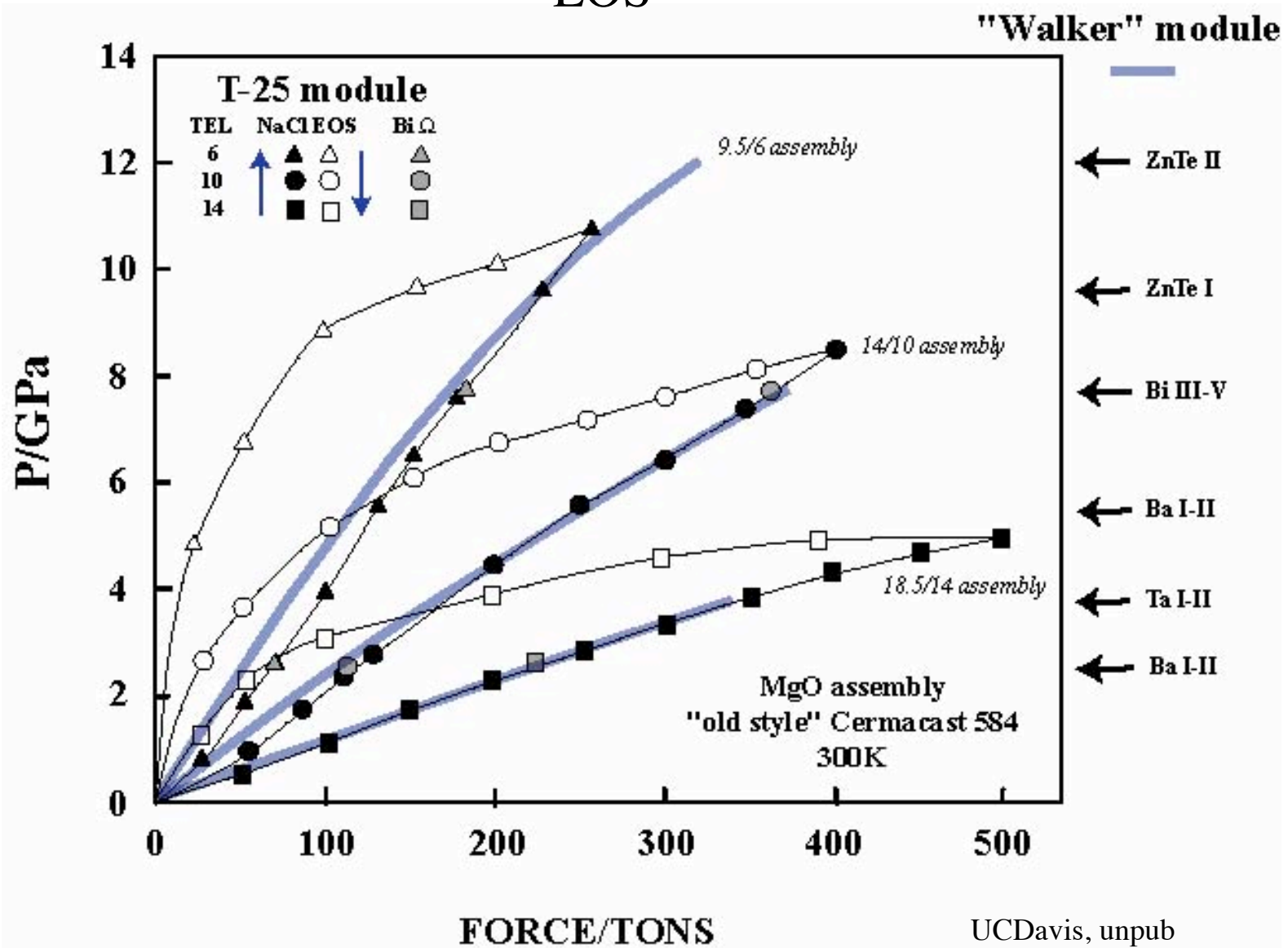


Fig. 4. High-temperature pressure calibration curves are shown (black lines). Room temperature calibrations (grey) plot below the high temperature curves of each assembly. The transformations on which the calibrations are based are CaGeO_3 garnet to perovskite (gt-pv), SiO_2 coesite to stishovite (co-stish), Mg_2SiO_4 forsterite to wadsleyite (α - β) and wadsleyite to ringwoodite (β - γ), and MgSiO_3 majorite to perovskite (Mj-Pv). The occurrence of either the high or low-pressure phase at each transformation is indicated by either filled or opened circles respectively.

EOS



Let the fun begin!